

IDENTIFICATION DES MODÈLES DE RUPTURE DUCTILE TENANT COMPTE DE L'ÉTAT DE CONTRAINTE, DE LA VITESSE DE DÉFORMATION ET DE LA TEMPÉRATURE

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Modeling Approaches

Option #3

Non-porous
plasticity



Damage
indicator model

Option #2

Porous
plasticity



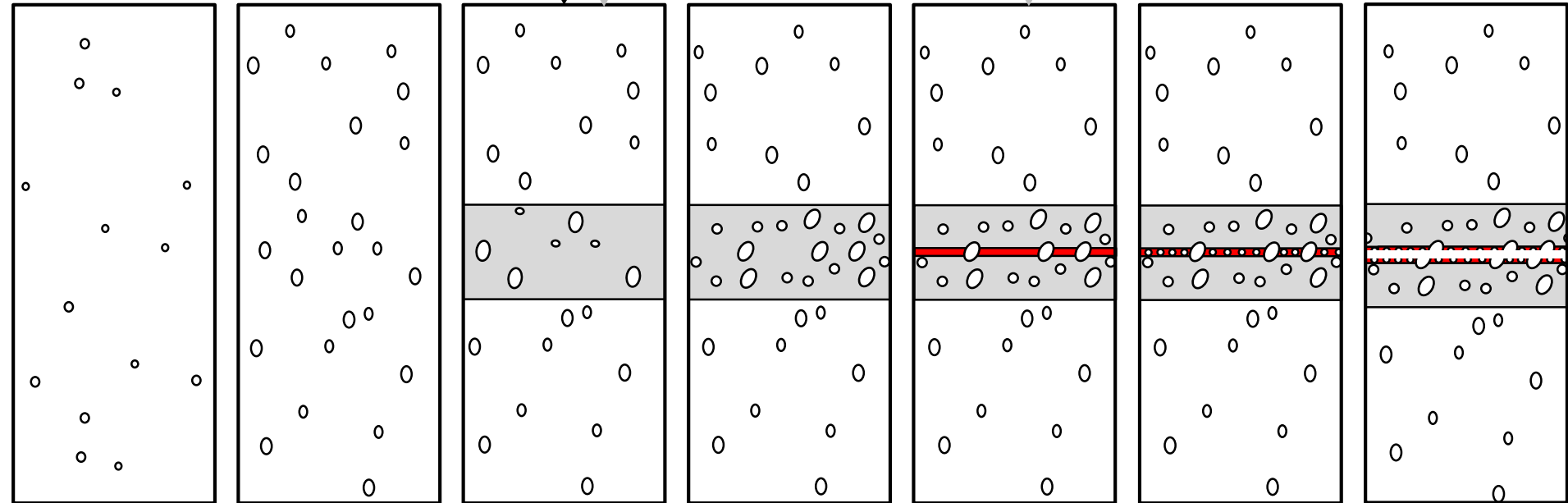
Localization
criterion

Option #1

Porous
plasticity

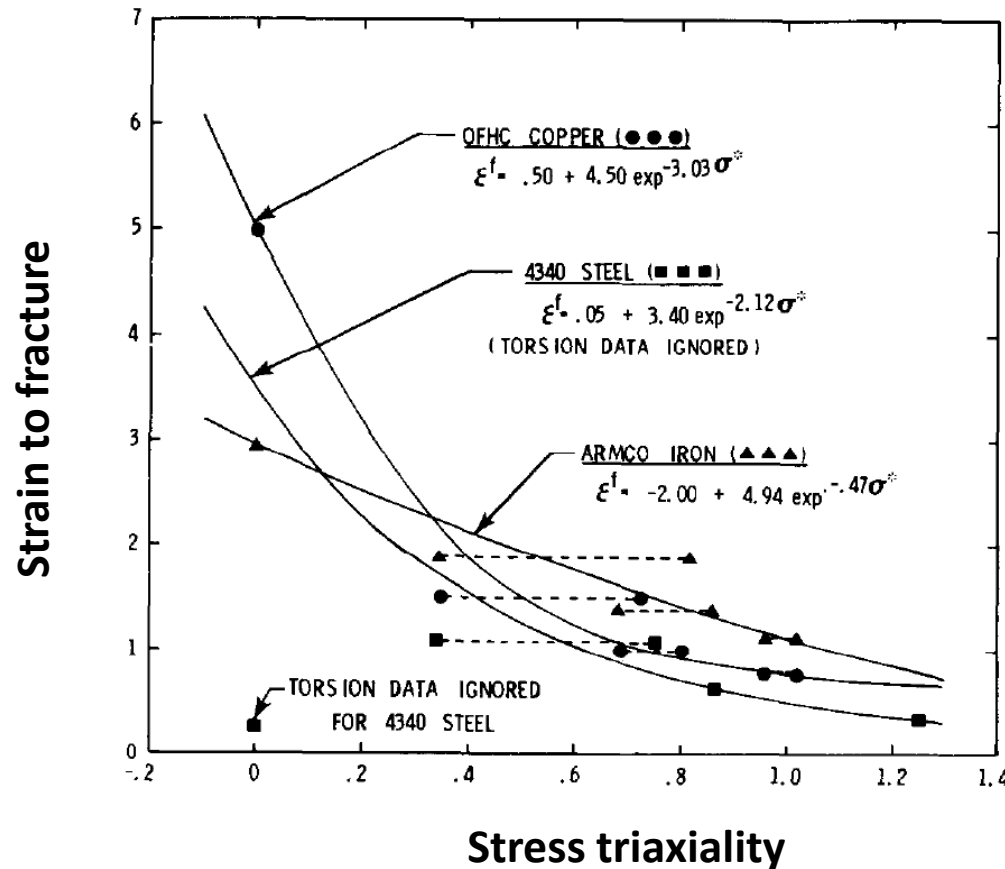


Coalescence
criterion



ETH Option #3: Damage Indicator Approach

Chose stress-state dependent fracture criterion such as Cockcroft & Latham (1968), Johnson & Cook (1985), Bai & Wierzbicki (2010), etc



$$\bar{\epsilon}_f^{pr} = \bar{\epsilon}_f^{pr} [\eta, \bar{\theta}]$$

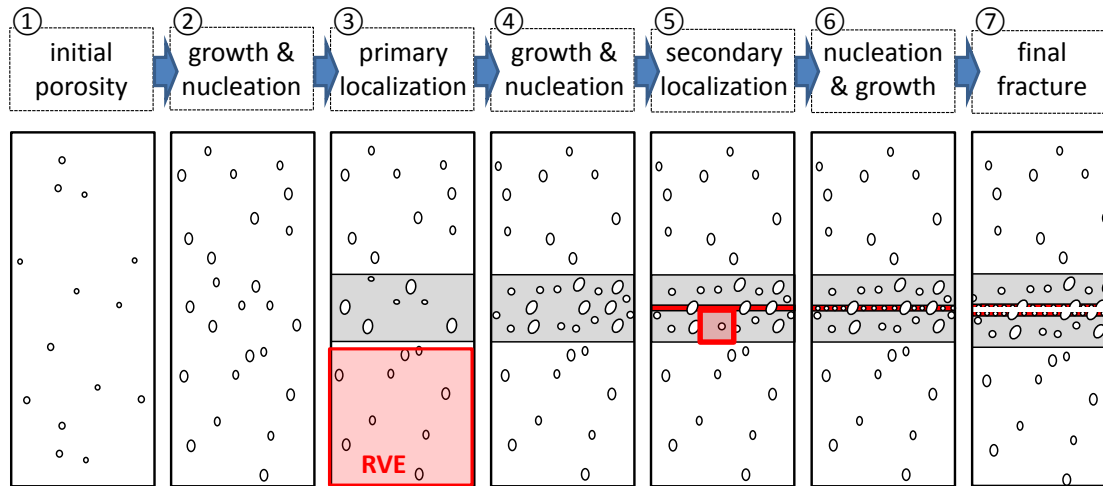


$$D = \int \frac{d\bar{\epsilon}_p}{\bar{\epsilon}_f^{pr} [\eta, \bar{\theta}]}$$

$$D = 0 \quad (\text{initial})$$

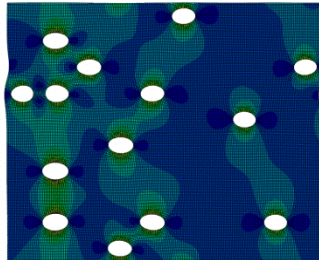
$$D = 1 \quad (\text{fracture})$$

Localization Analysis



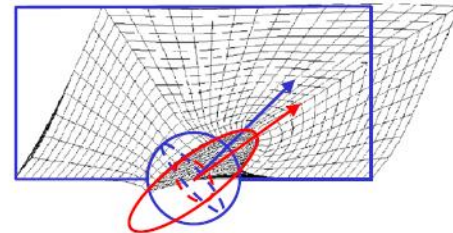
Macroscopic Localization Analysis

- Using Phenomenological porous plasticity
→ Nahshon & Hutchinson (2008)
- Using Homogenization based porous plasticity
→ Danas & Ponte Castaneda (2012)
- Unit cell with random void distribution
→ ongoing



Coalescence Analysis

- **Single void unit cell analysis**
→ Tvergaard, Pardoen, Needleman, Faleskog, Hopperstad & others
→
- Analytical criteria
→ e.g. Leblond, Benzerga & others



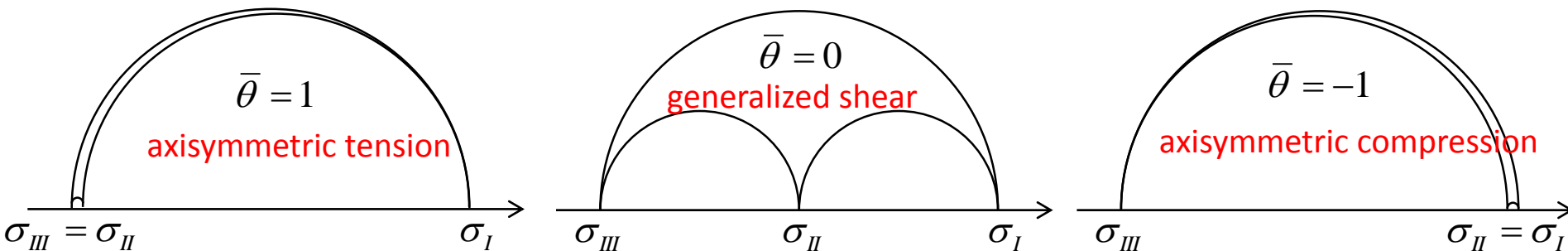
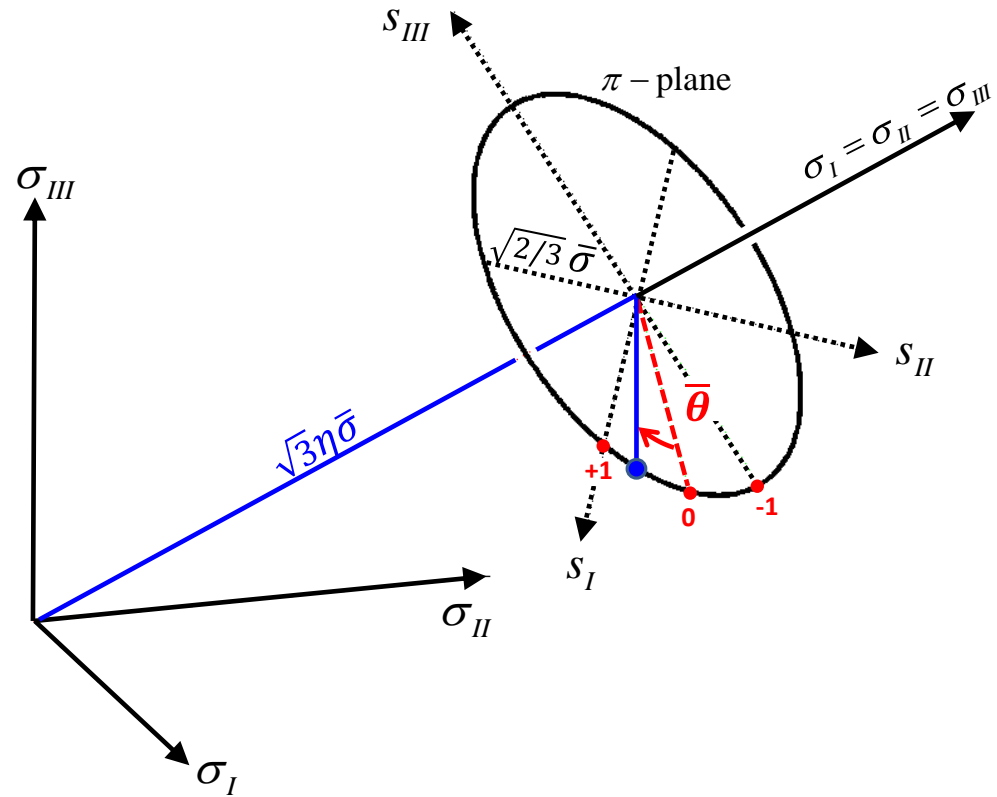
Definition: Lode angle parameter

- Lode number (Lode, 1926)

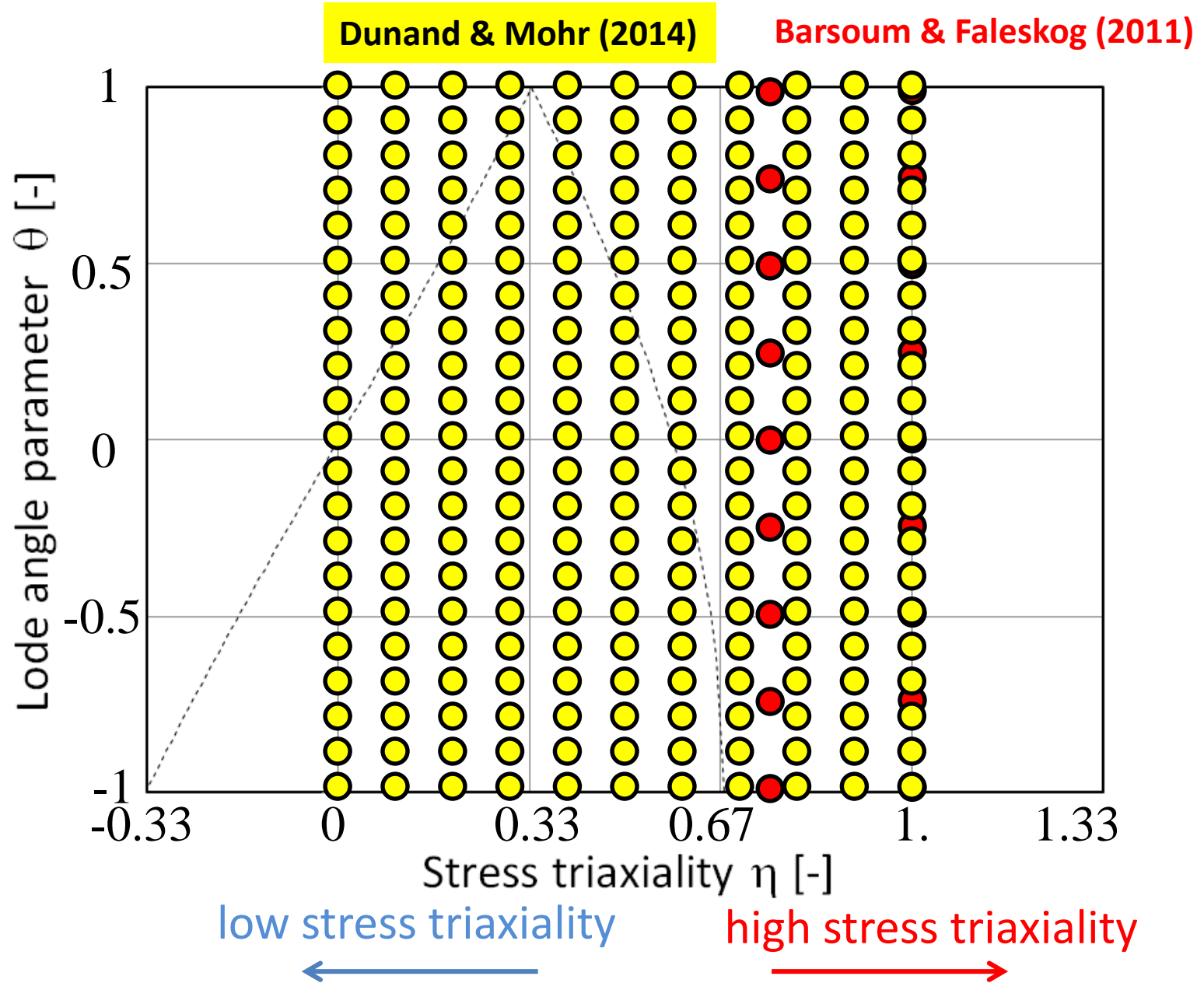
$$\mu = \frac{2\sigma_{II} - \sigma_I - \sigma_{III}}{\sigma_I - \sigma_{III}}$$

- Lode angle parameter

$$\bar{\theta} = 1 - \frac{2}{\pi} \arccos(\xi) \cong -\mu$$

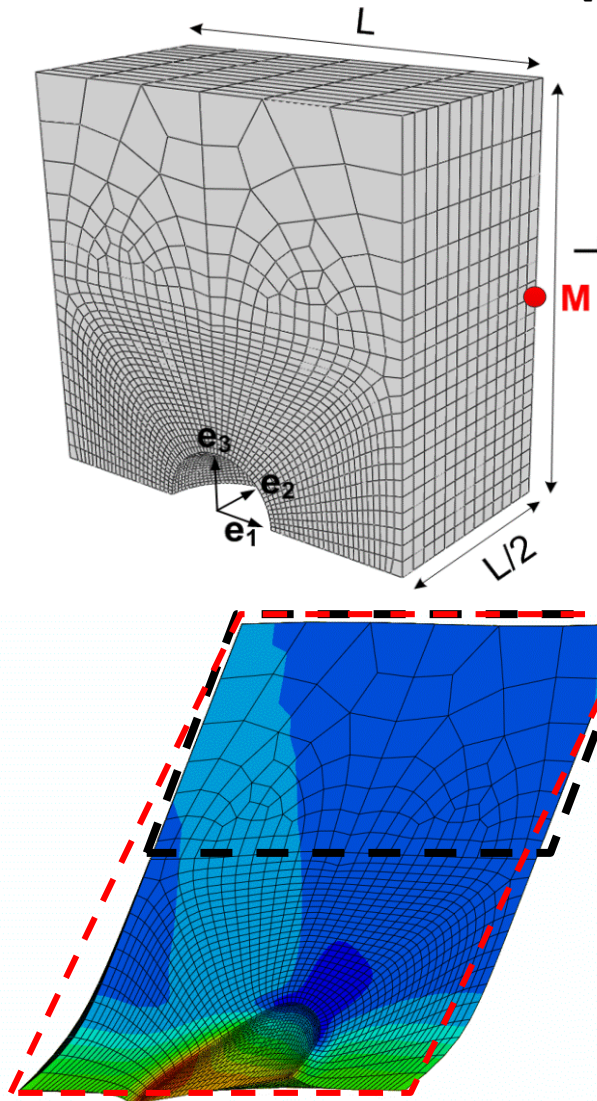


Stress State Map: Unit Cell Analysis



Unit Cell Model

[Dunand and Mohr, JMPS 2014]



- Matrix material: von Mises with isotropic hardening
- Initial defect volume fraction: 1.2%
- Periodic boundary conditions
- **Macroscopic stress triaxiality and Lode parameter kept constant throughout loading**
- Kinematic criterion to detect localization

F Deformation gradient of the cell

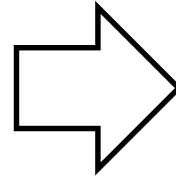
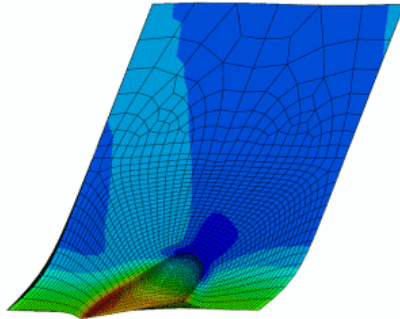
F^0 Deformation gradient outside band of localization

$$\xi = \frac{\|\dot{F}\|}{\|\dot{F}^0\|} \gg 1$$

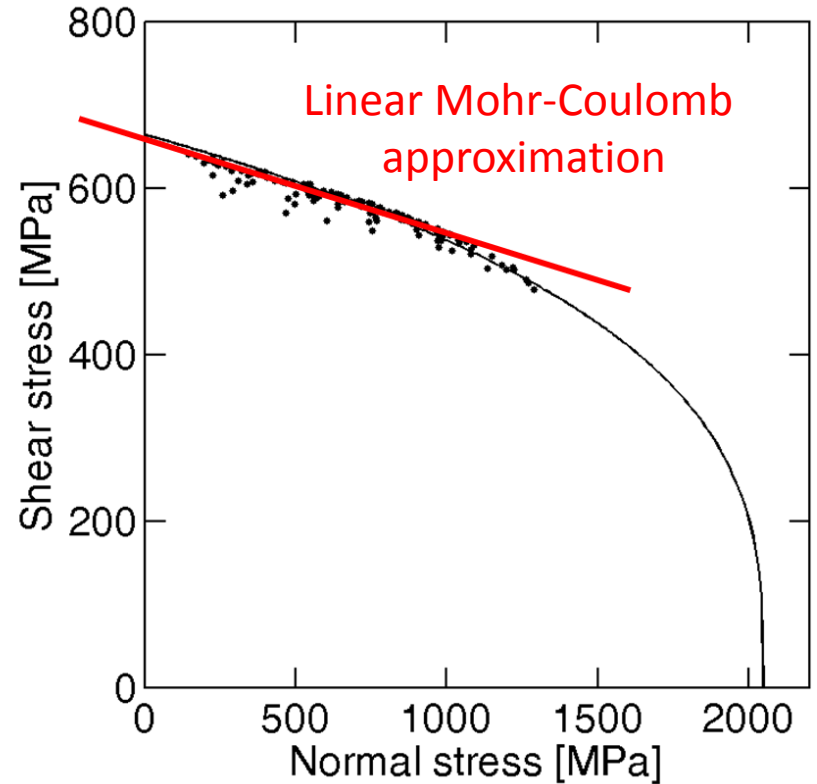
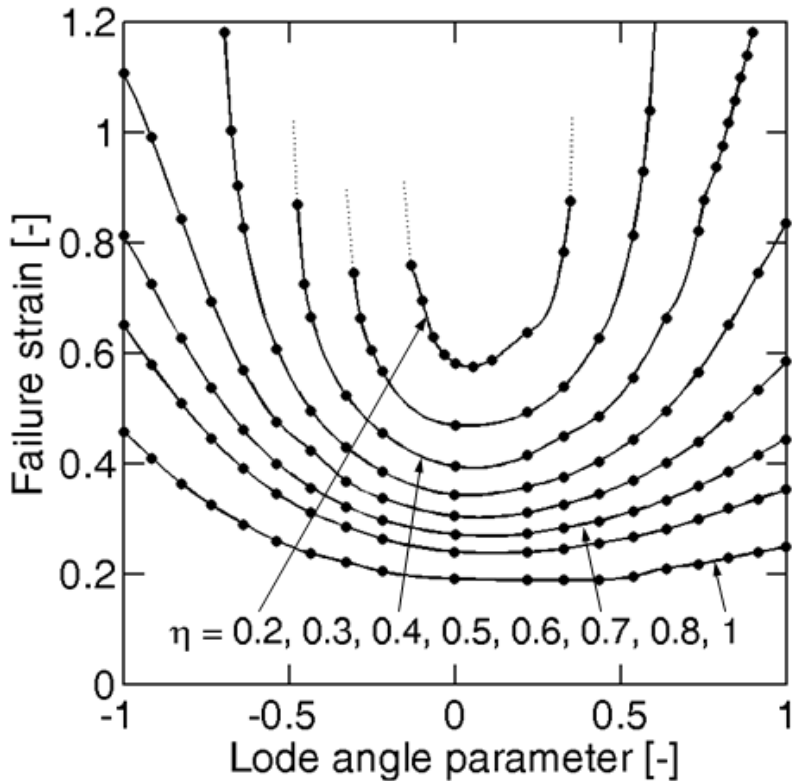
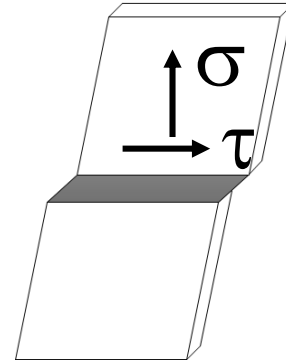
Needleman & Tvergaard (1992)

[Dunand and Mohr, JMPS 2014]

Unit Cell with Central Void



Stresses on Plane of Localization



Hosford-Coulomb model

[Mohr and Marcadet, IJSS 2015]

Coordinate transformation

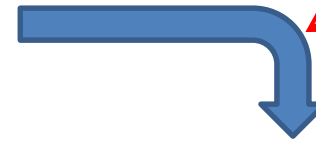


Principal stress space $\{\sigma_I, \sigma_{II}, \sigma_{III}\}$

$$\bar{\sigma} = h(\bar{\theta}, \eta)$$

Haigh-Westergaard space $\{\eta, \bar{\theta}, \bar{\sigma}\}$

Isotropic hardening law



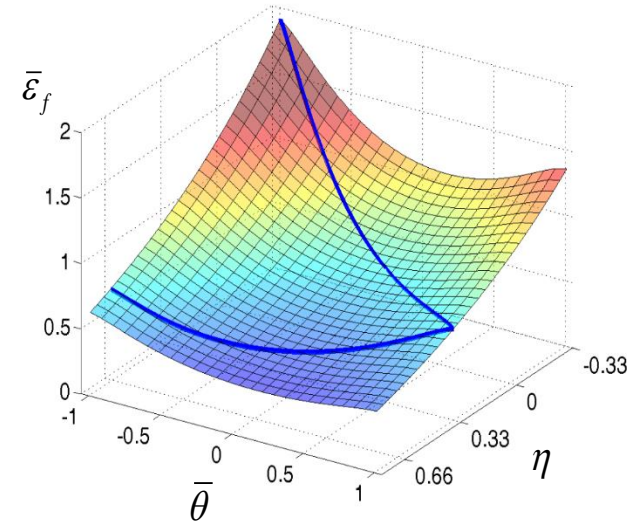
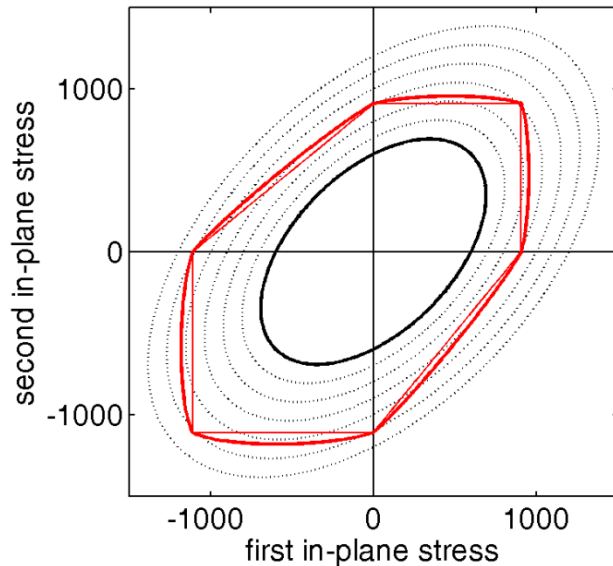
Mixed strain-stress space $\{\eta, \bar{\theta}, \bar{\varepsilon}_p\}$

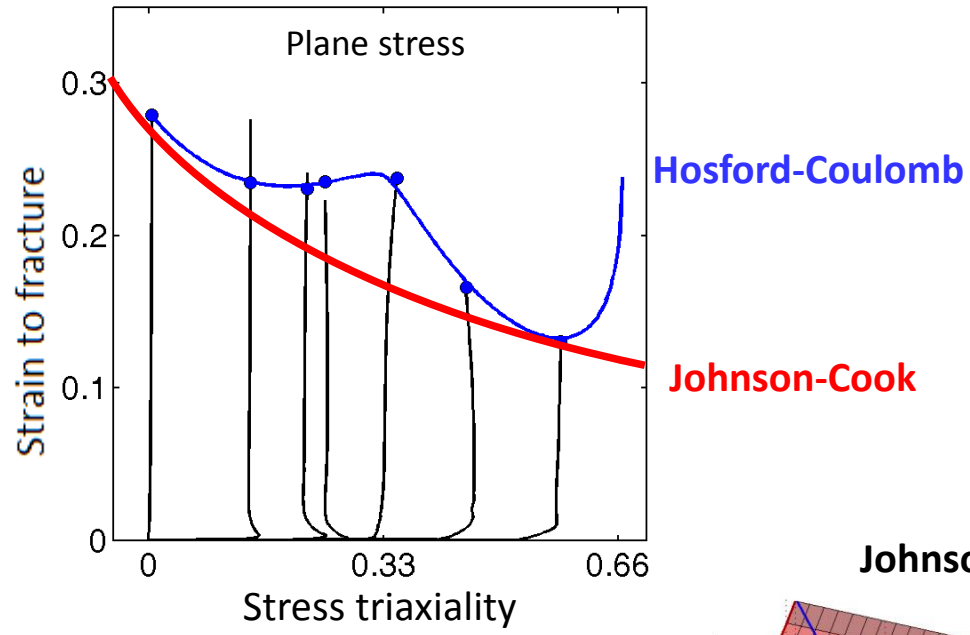
$$\bar{\varepsilon}_p = f(\eta, \bar{\theta})$$

Fracture initiation criterion

Hosford-Coulomb

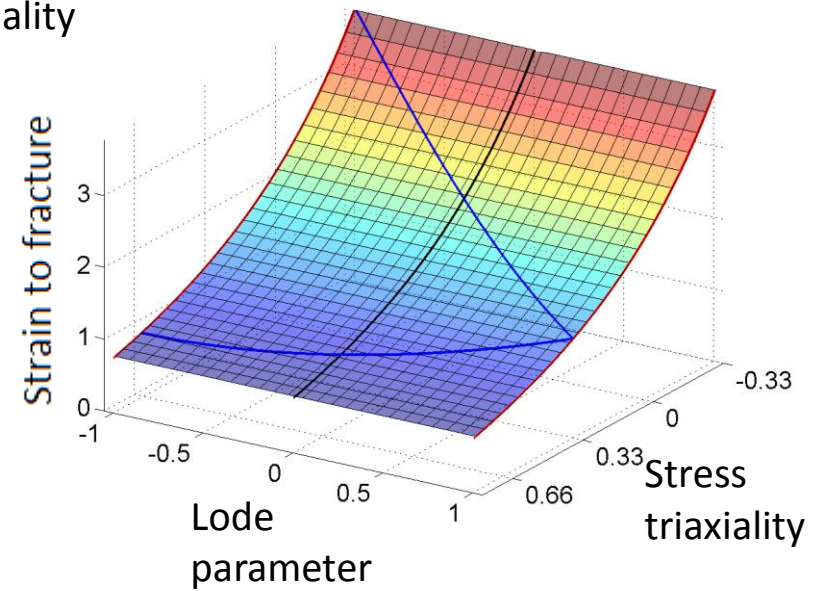
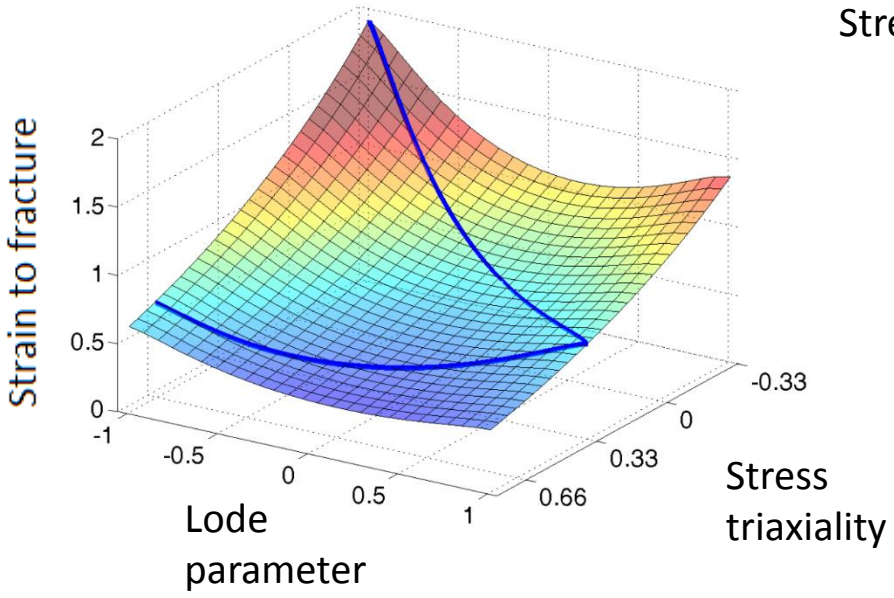
$$\bar{\sigma}_{Hf} + c[\sigma_I + \sigma_{III}] = b$$



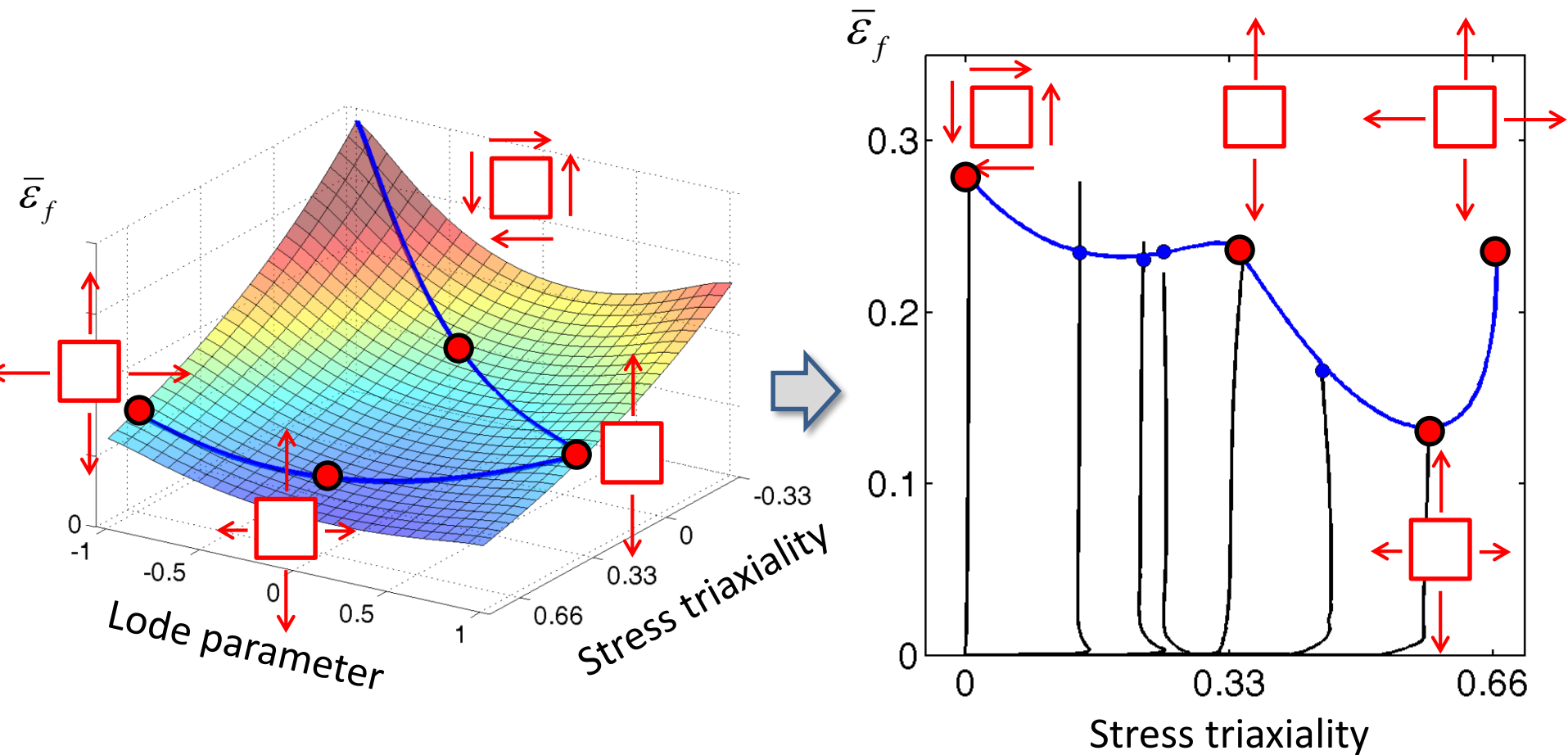


Hosford-Coulomb

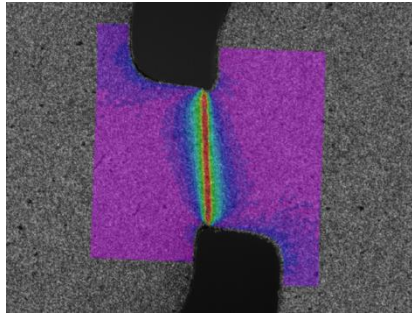
Johnson-Cook



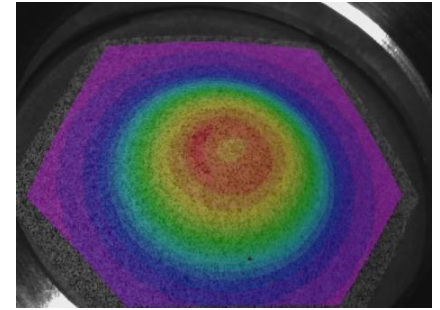
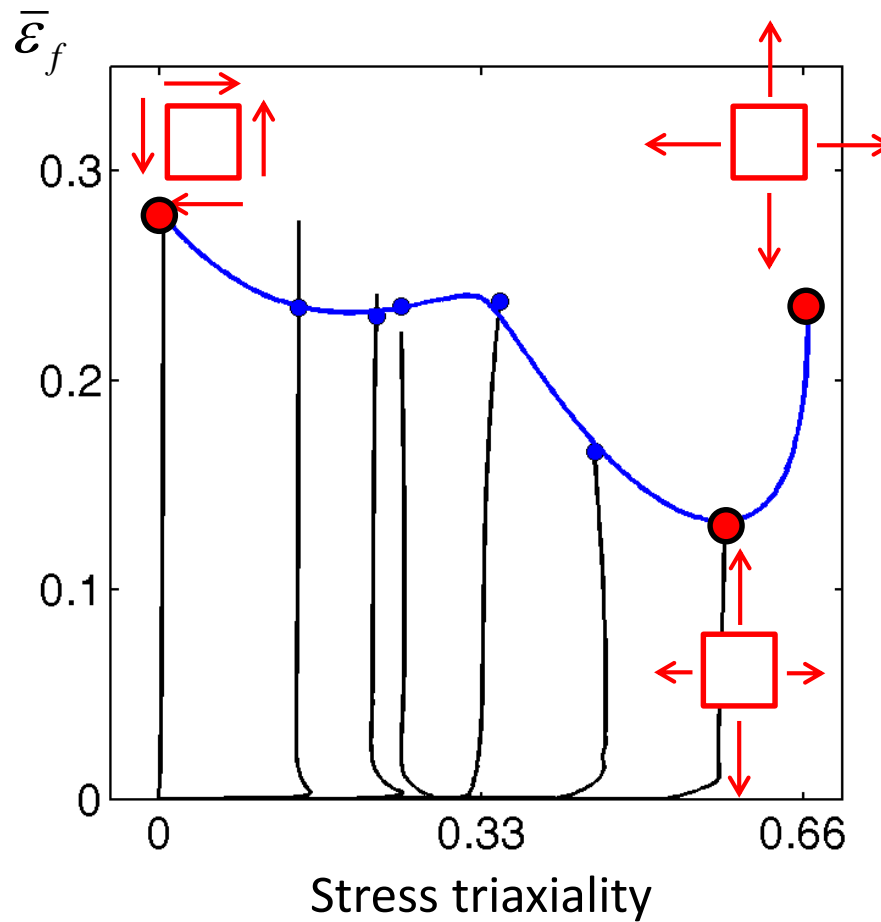
Important Points for Experimental Characterization



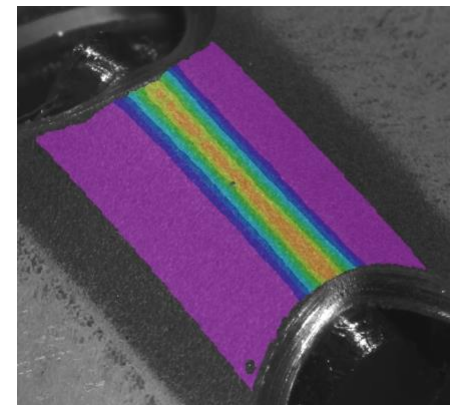
ETH Experiments with Proportional Loading Paths



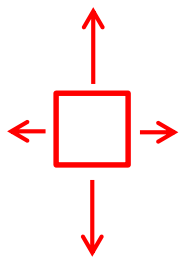
In-plane shear



Mini-punch

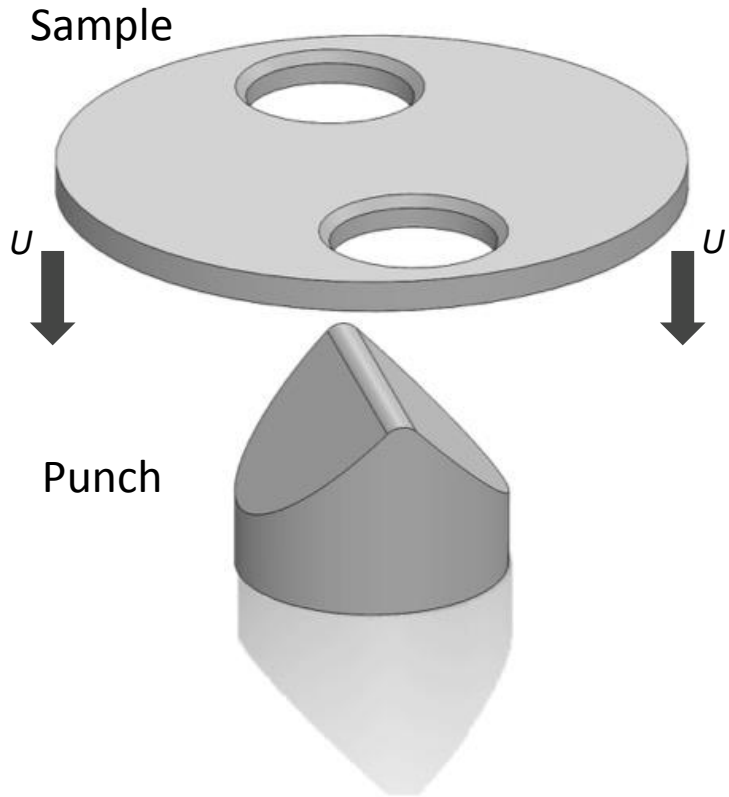


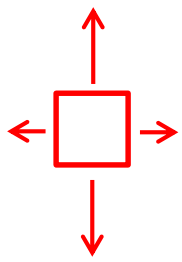
Mini-Nakazima



Plane strain tension: Mini-Nakazima with Dihedral Punch

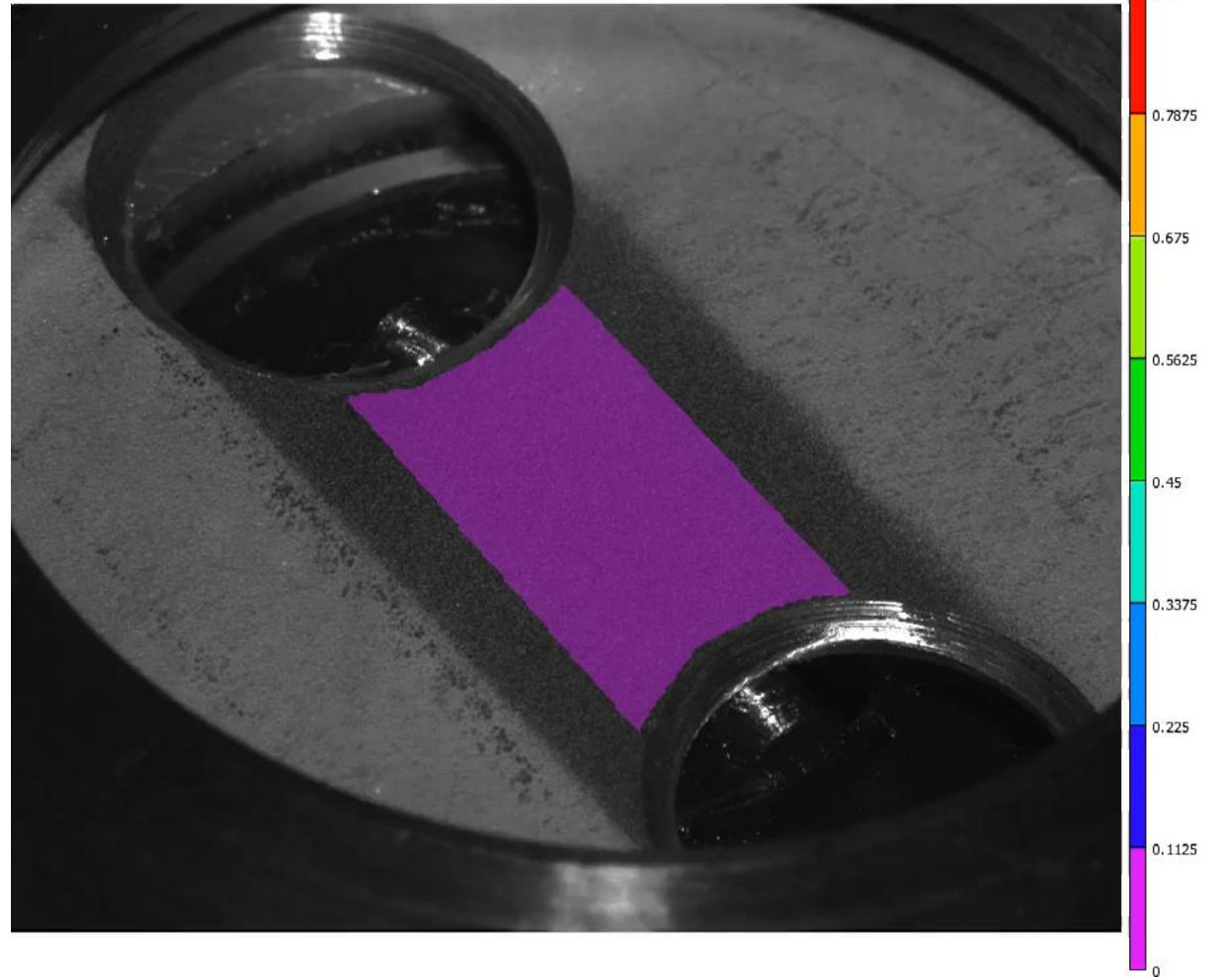
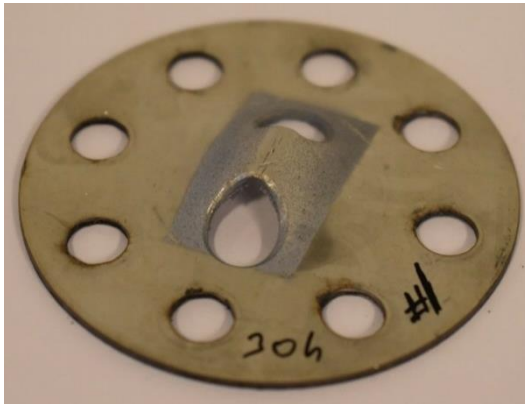
[Grolleau et al., IJMS 2019]





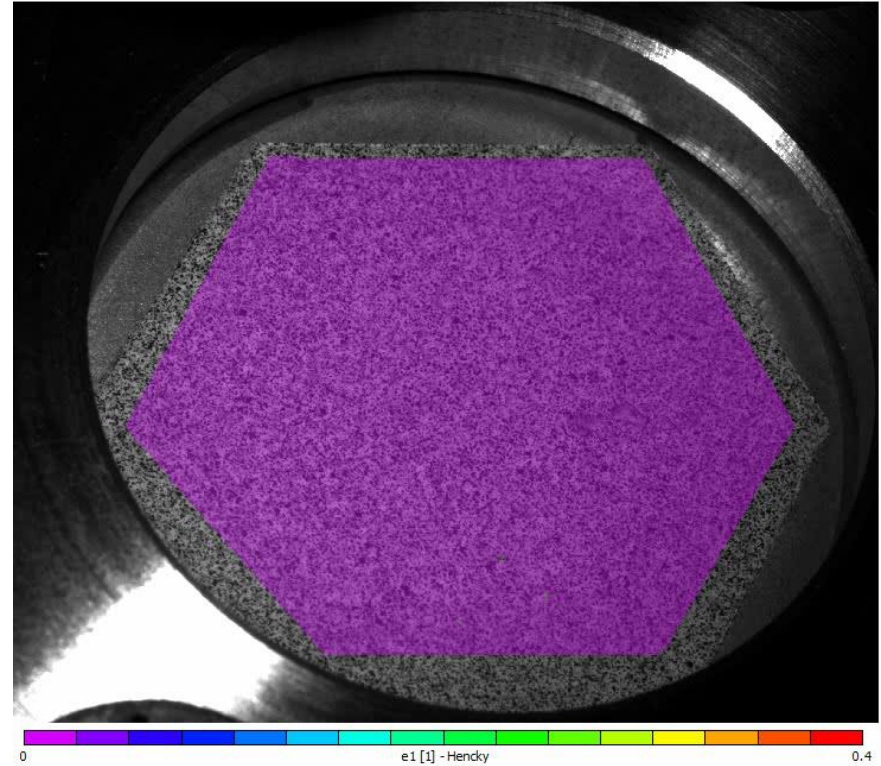
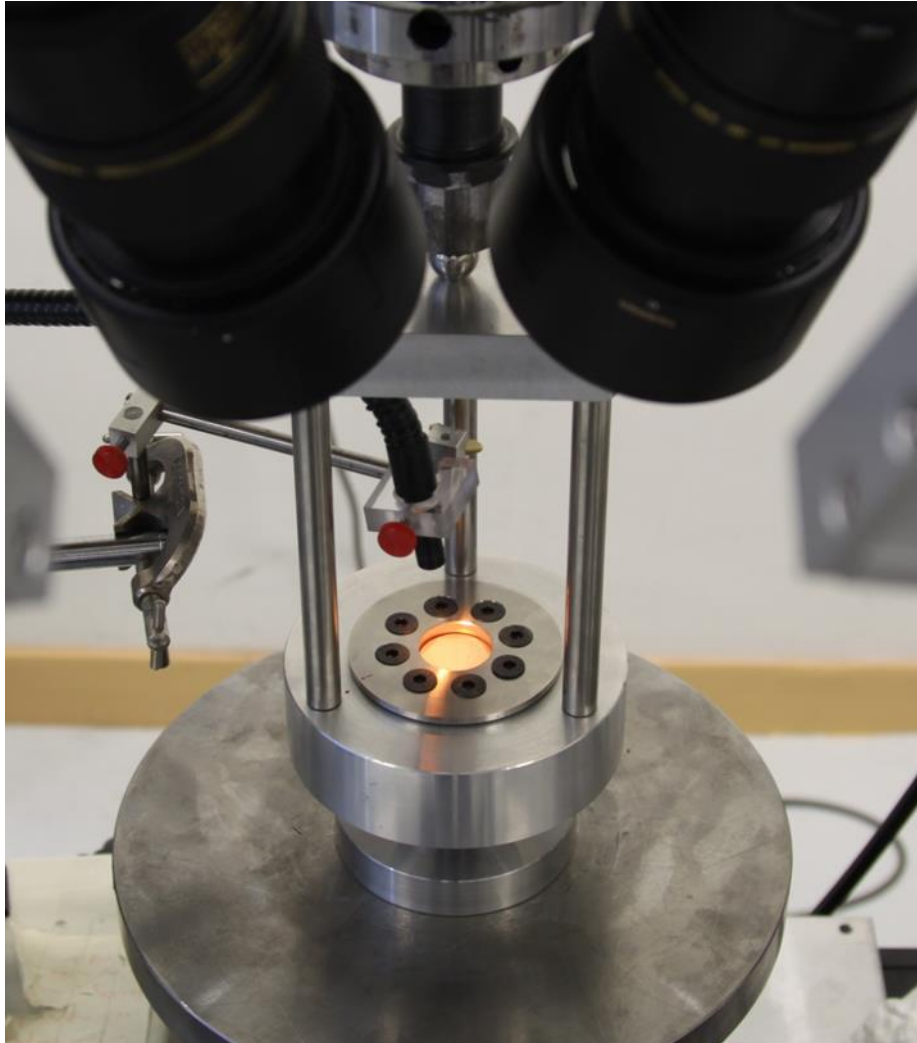
Plane strain tension: Mini-Nakazima with Dihedral Punch

[Grolleau et al., IJMS 2019]



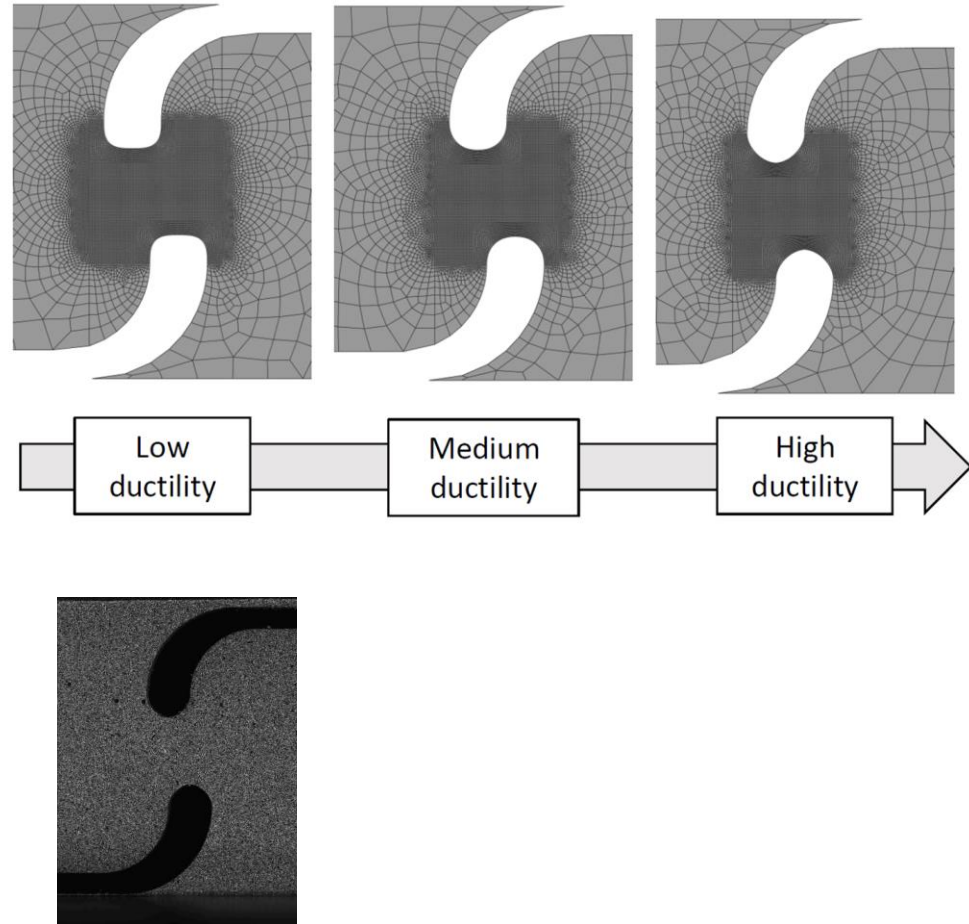
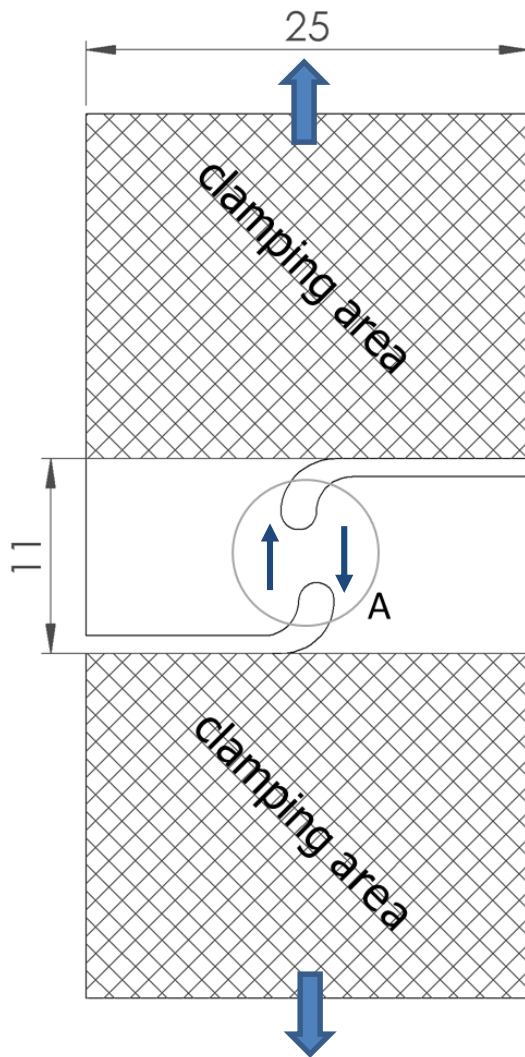
ETH Equi-biaxial tension: mini-punch test

[Roth & Mohr, IJP 2016]



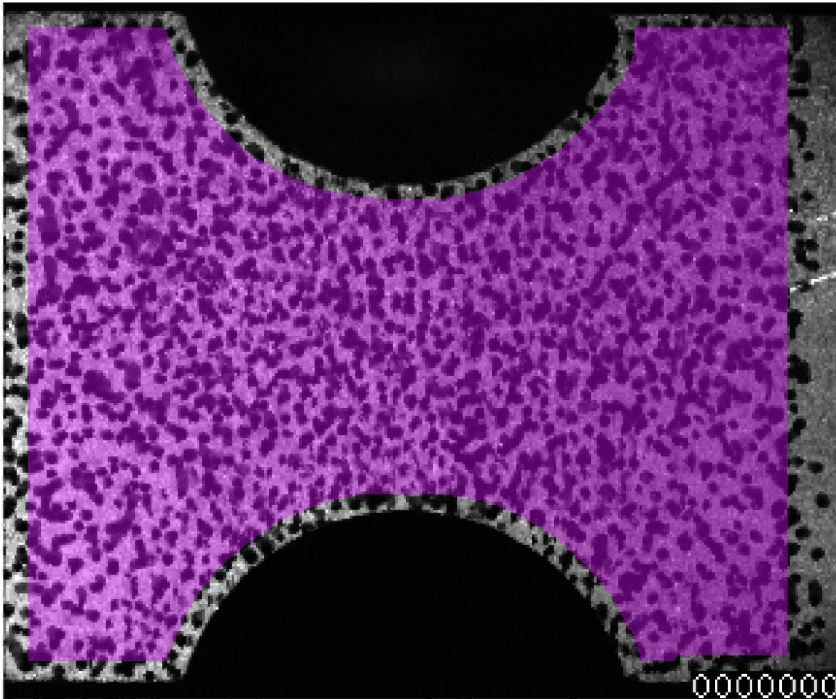
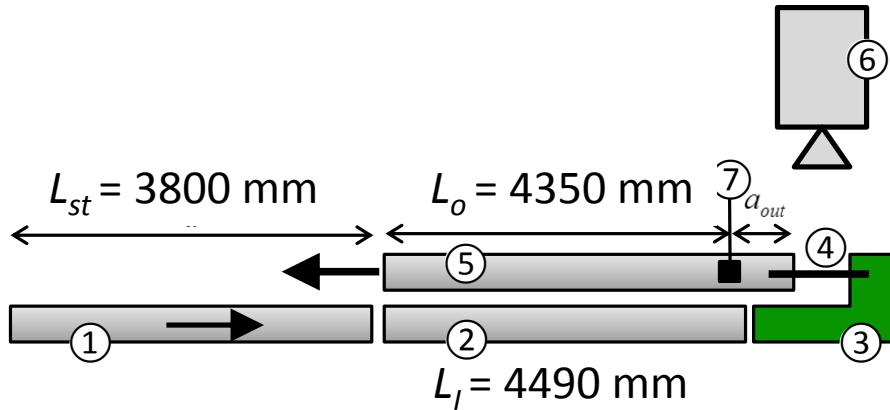
[Roth & Mohr, IJMS 2018]

- Optimal geometry depends on material ductility!

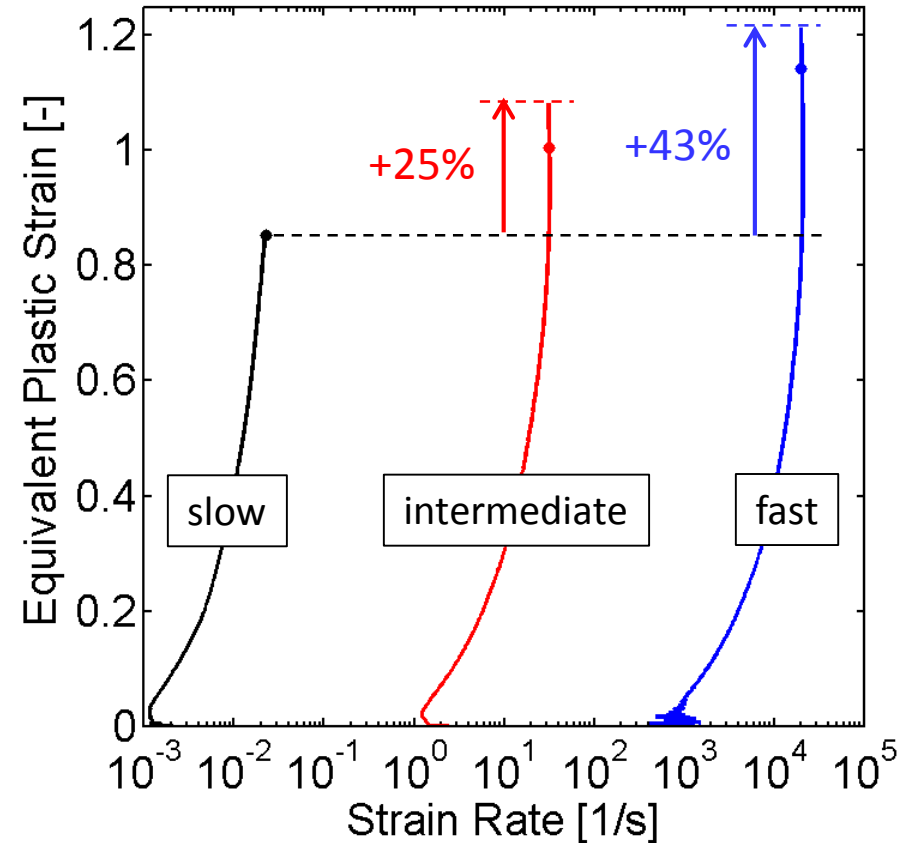


Effects of Strain Rate and Temperature on Ductile Fracture

SHB Tension experiments



Joint work with Prof. P. Forquin & G. Gary

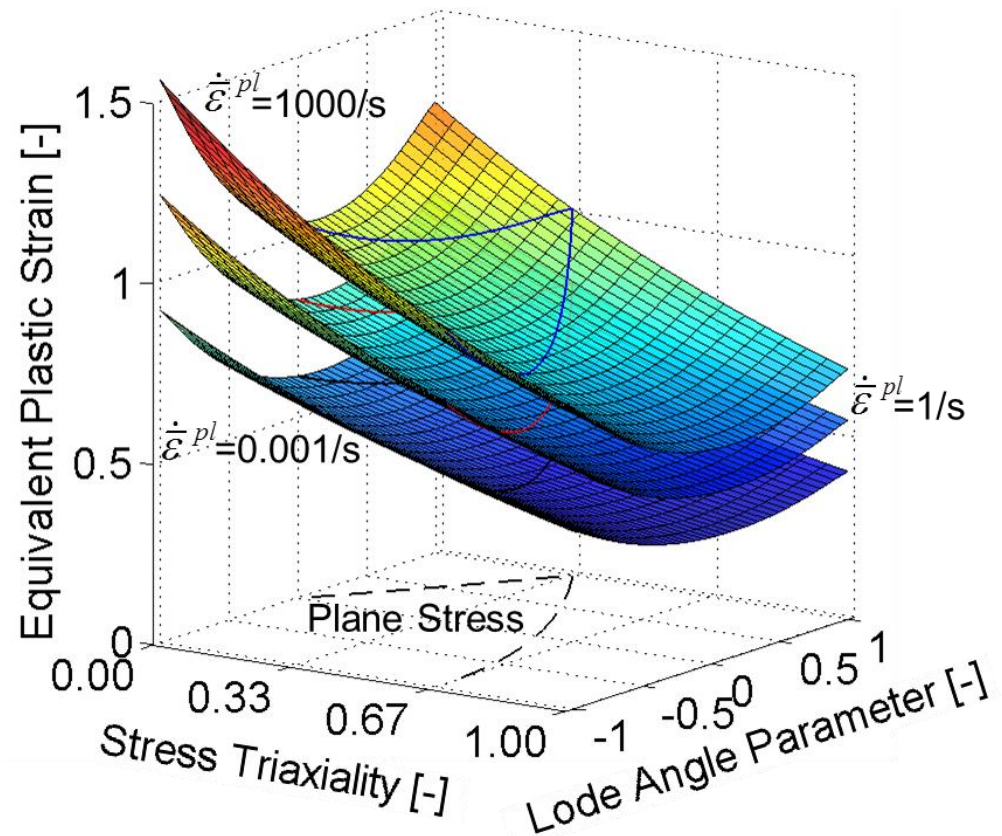


Strain to fracture increases
with strain rate!

Main assumption: strain to fracture increases with strain rate

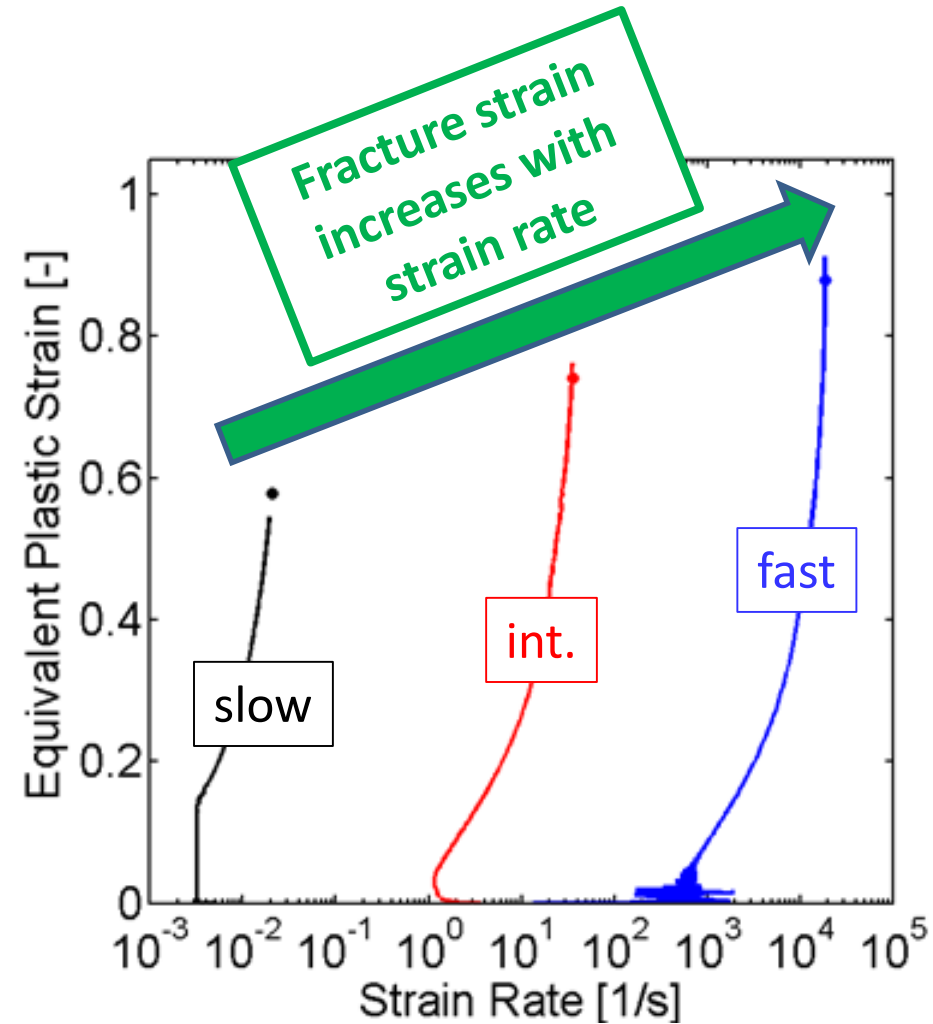
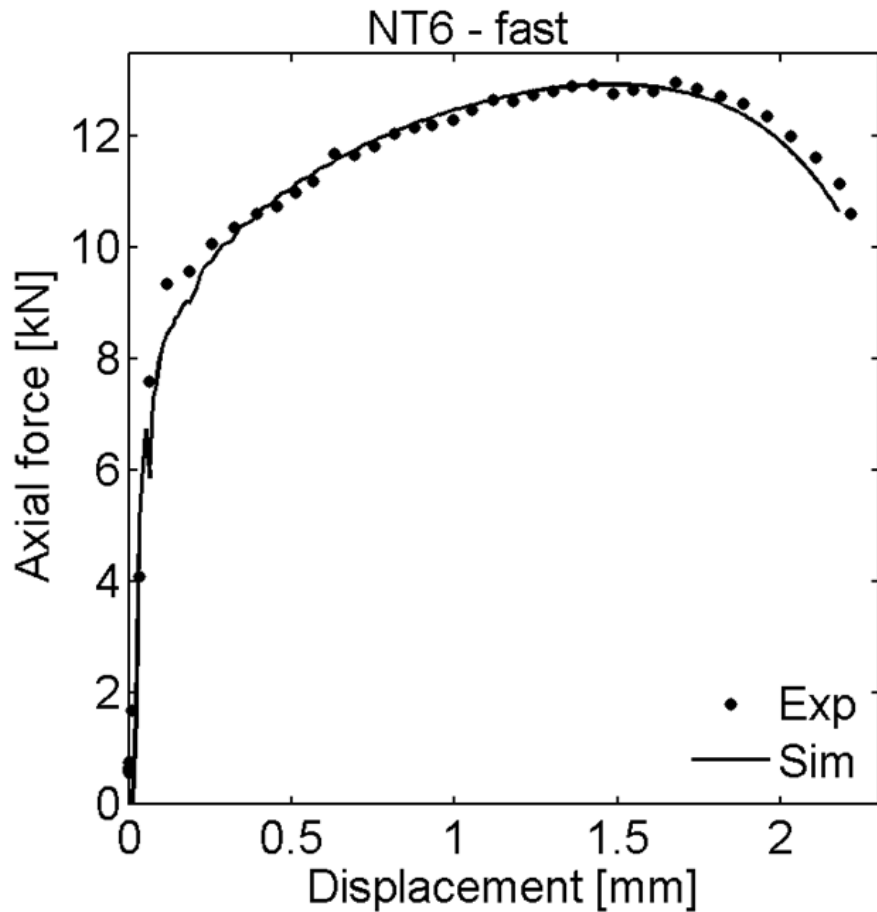
- Basis: Hosford-Coulomb Model

$$\bar{\varepsilon}_{f, RD} = \left(1 + \gamma \ln \left[\frac{\dot{\varepsilon}_P}{\dot{\varepsilon}_0} \right] \right) \bar{\varepsilon}_{f, RI}$$



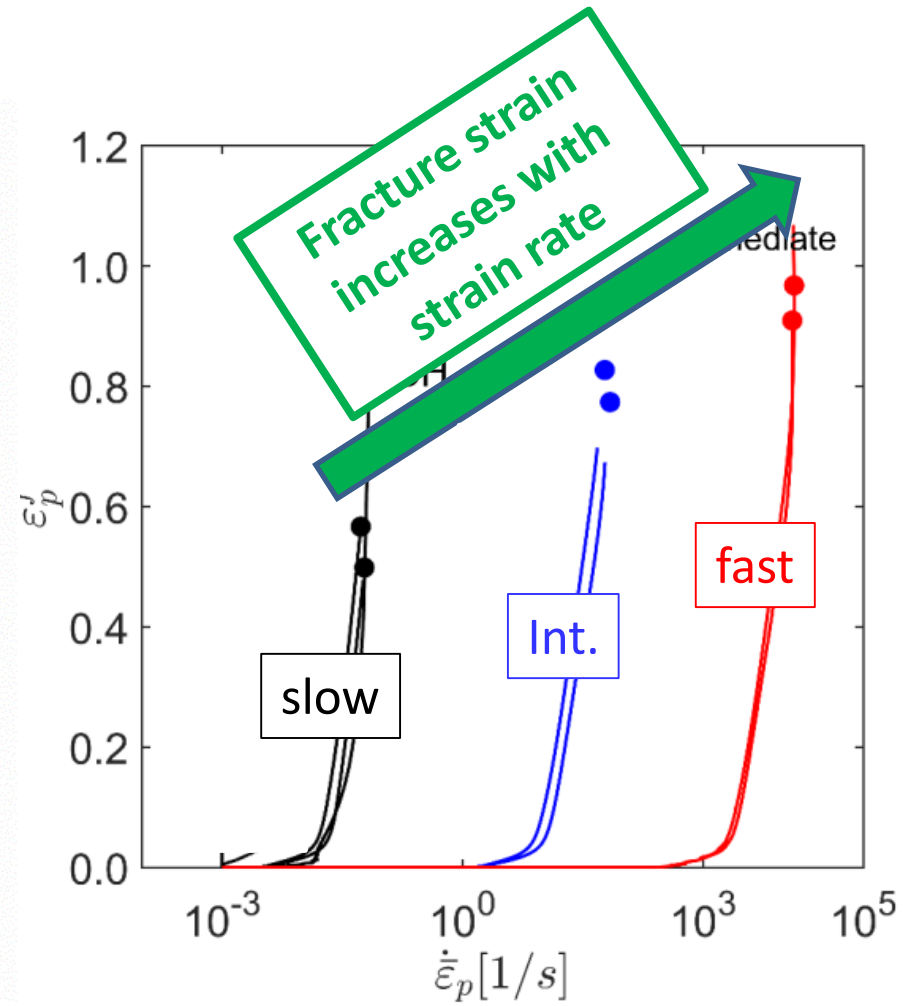
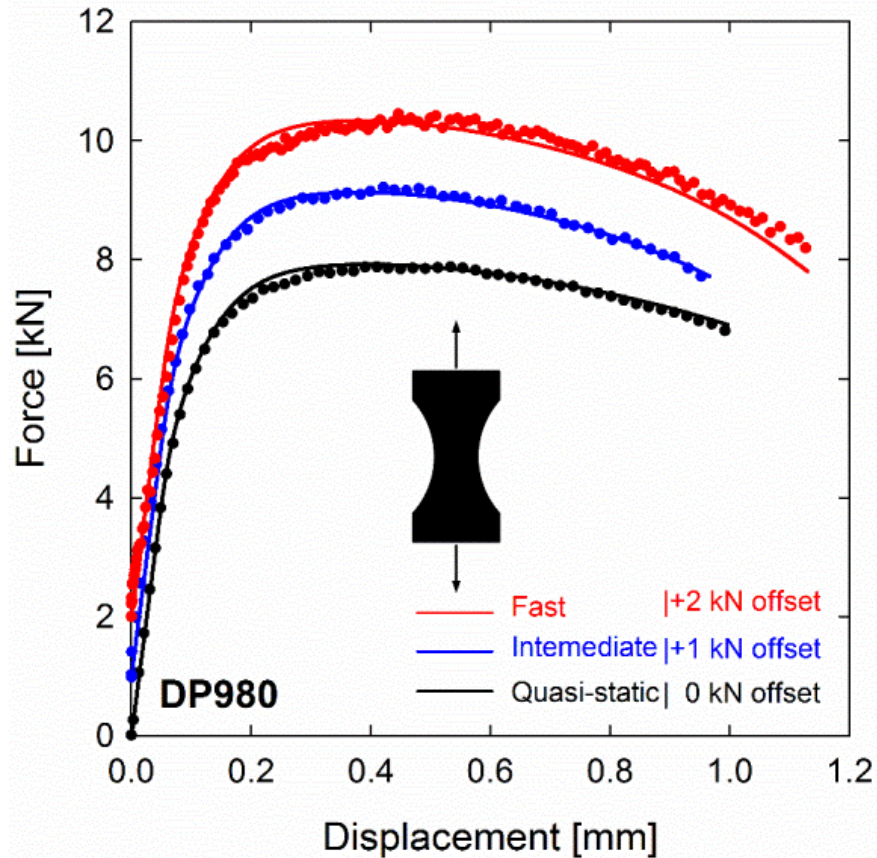
Confirmation for Material #1 (TRIP780)

Roth and Mohr (IJP, 2014)



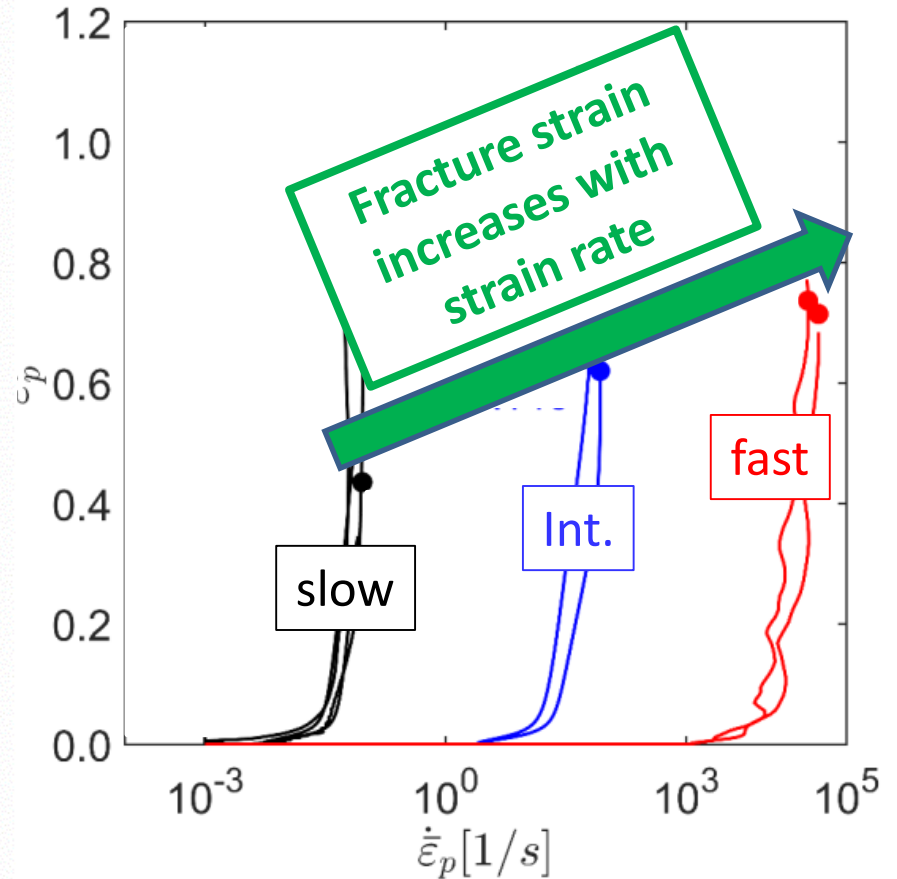
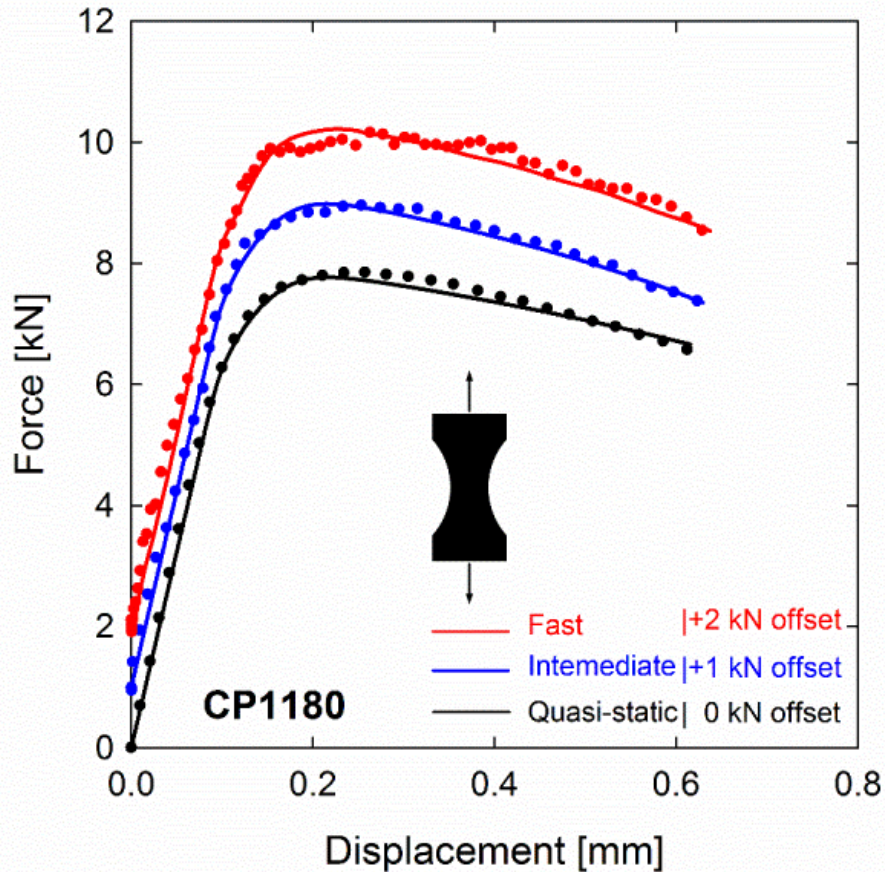
Confirmation for Material #2 (DP980)

Erice, Roth and Mohr (MOM, 2017)



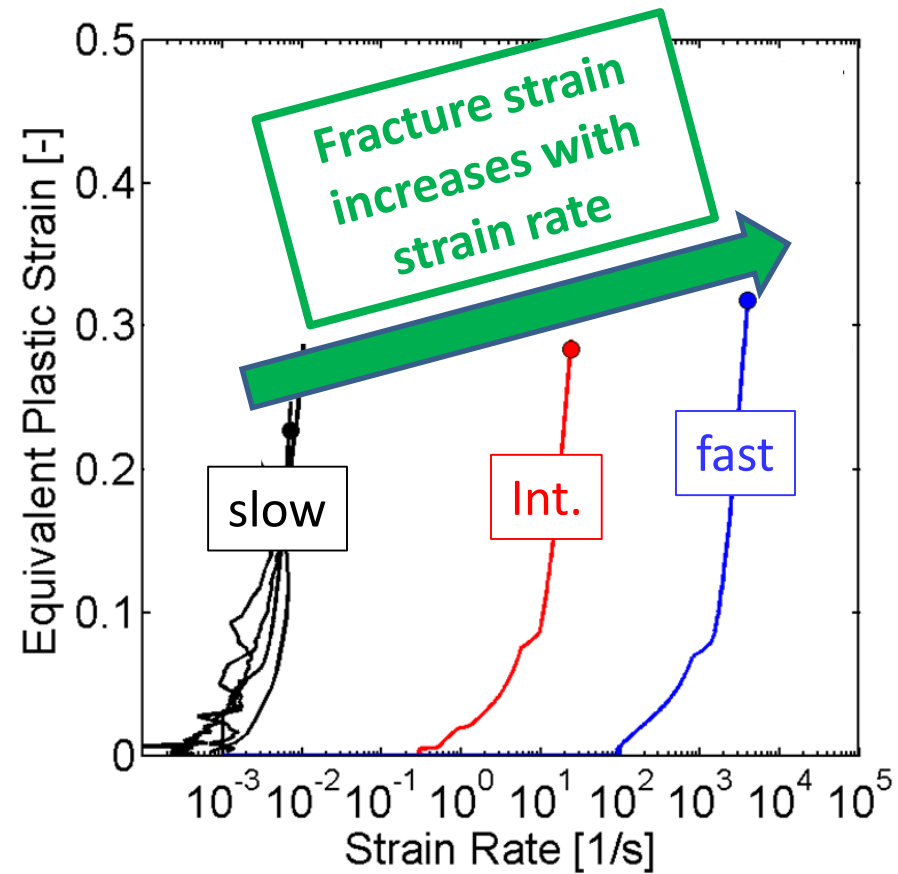
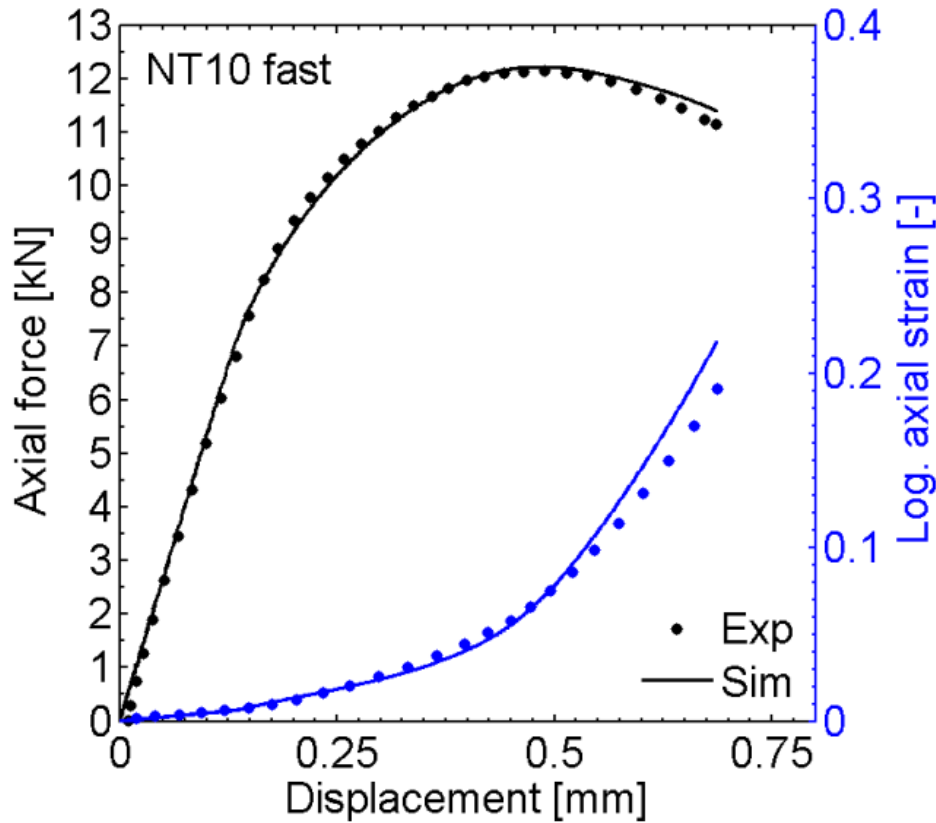
Confirmation for Material #3 (CP1180)

Erice, Roth and Mohr (MOM, 2017)



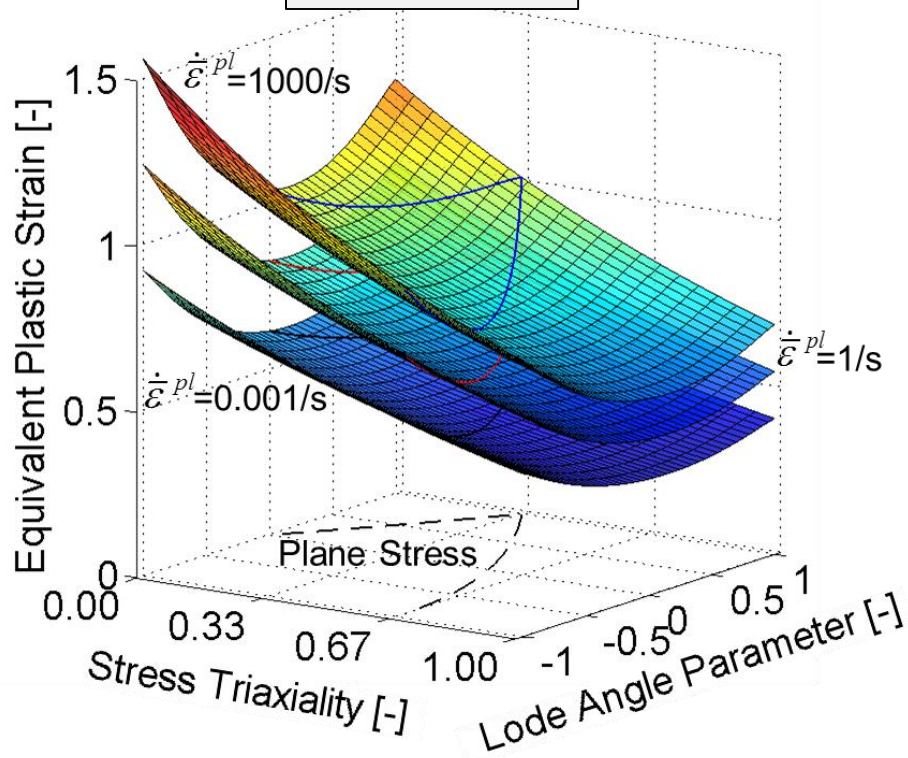
Confirmation for Material #4 (Mars300)

Fras, Roth and Mohr (IJIE, 2018)

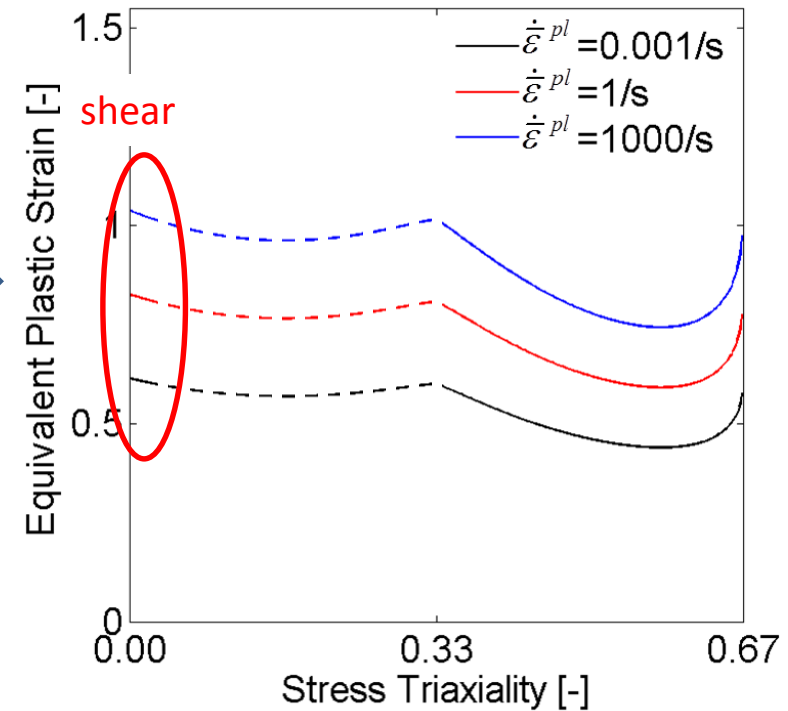


What is the effect of strain rate on fracture strain for pure shear fracture?

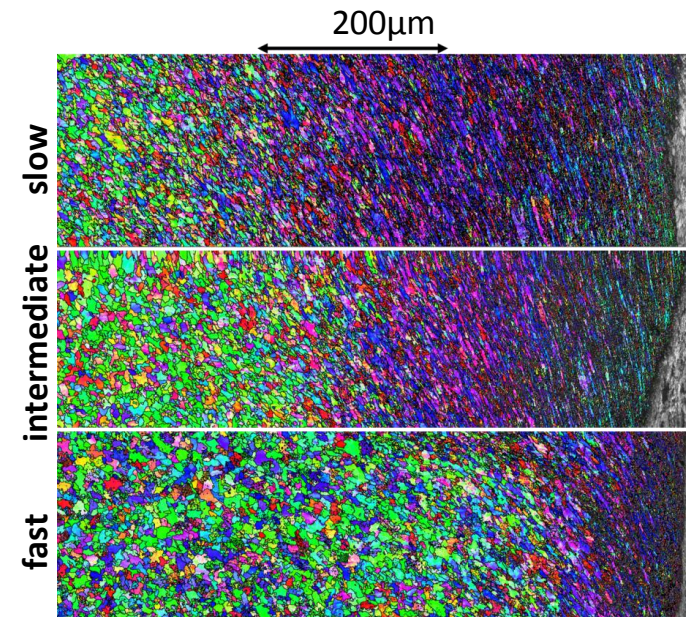
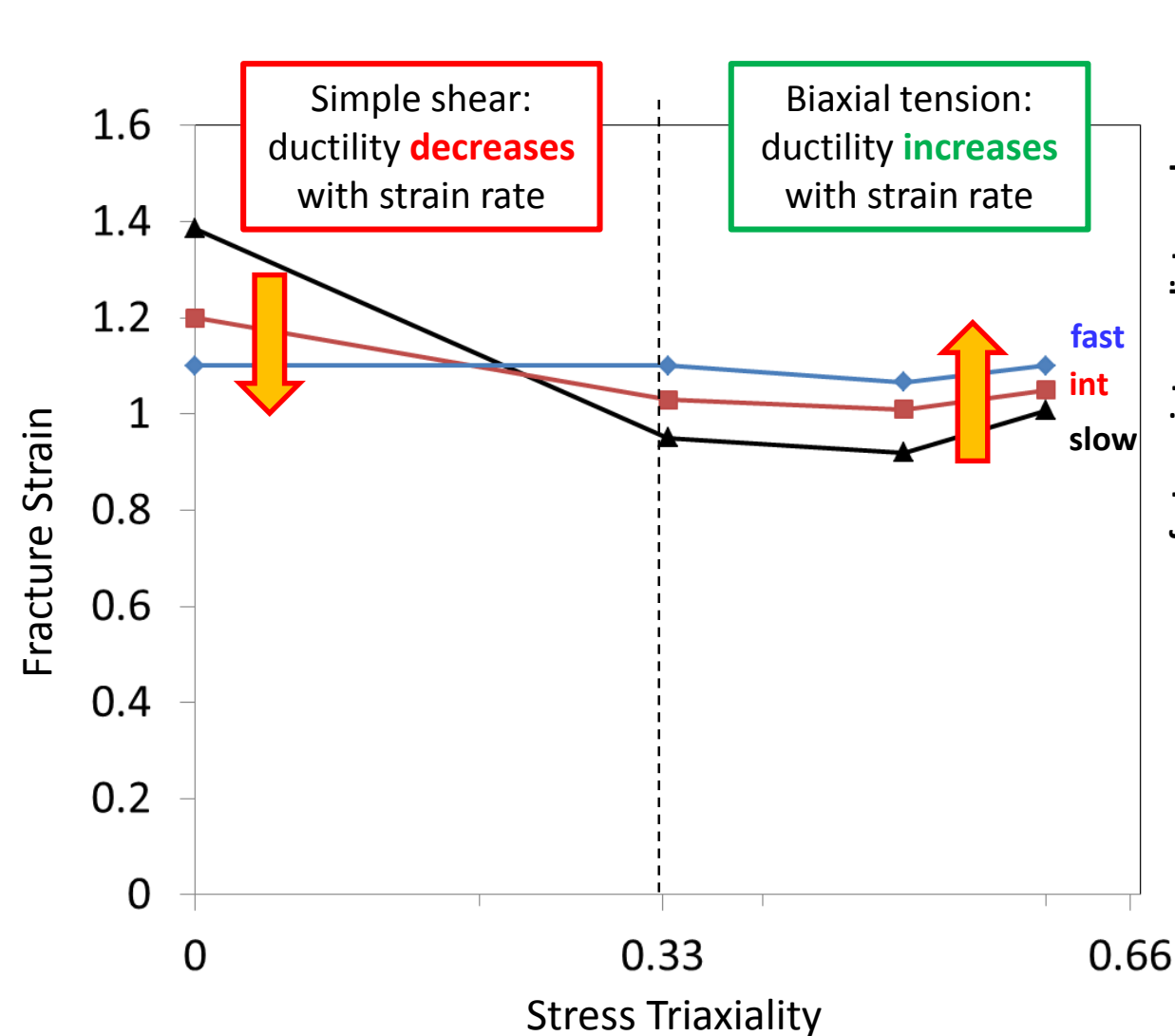
3D model



Plane stress model



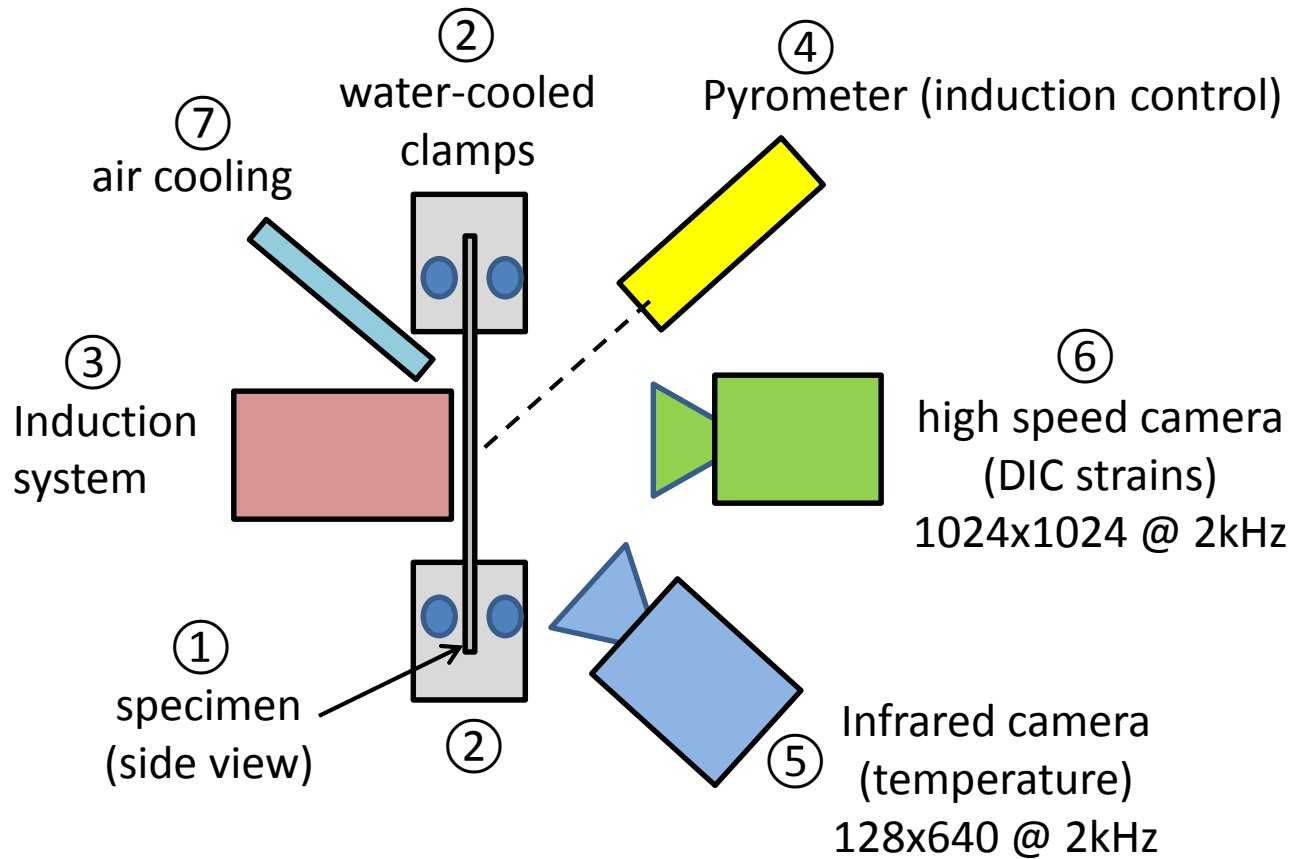
Effect of loading rate on fracture strain



Hypothesis:

Temperature softening effect overwrites increase in ductility due to high strain rate for shear

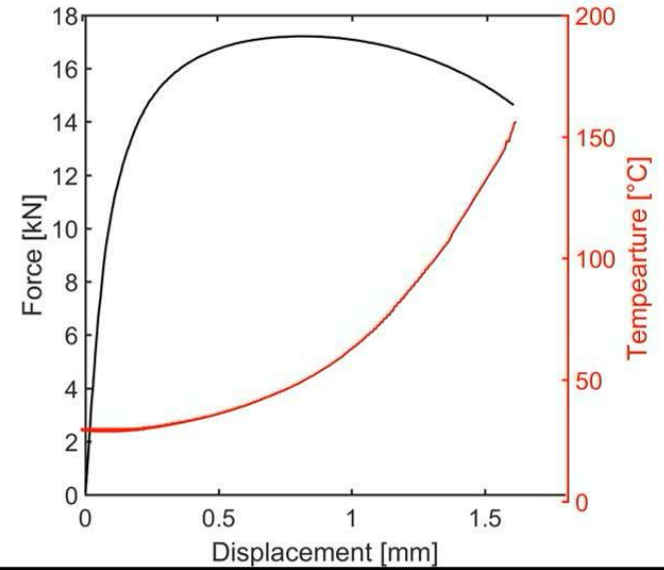
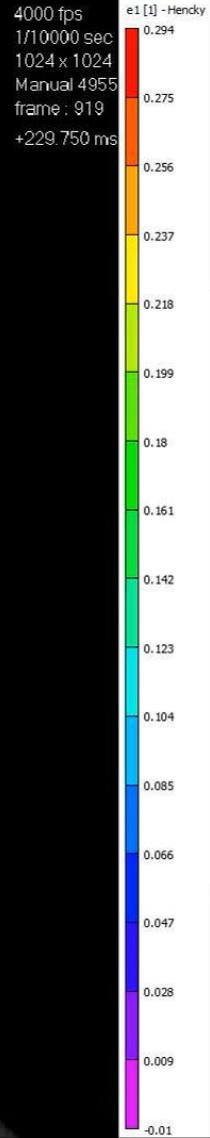
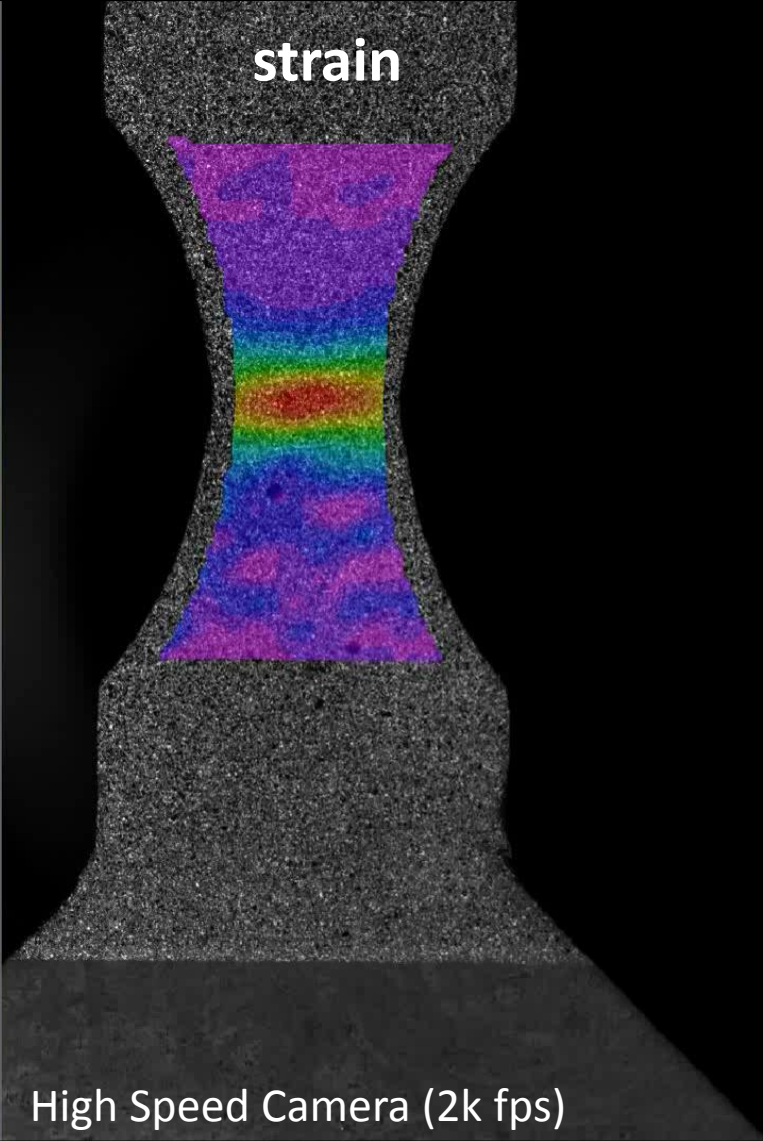
Experimental setup



temp.



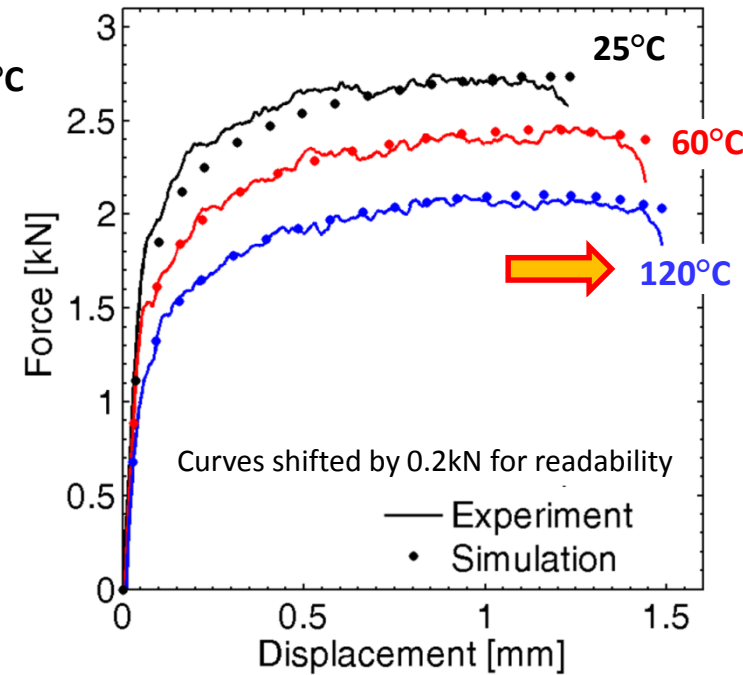
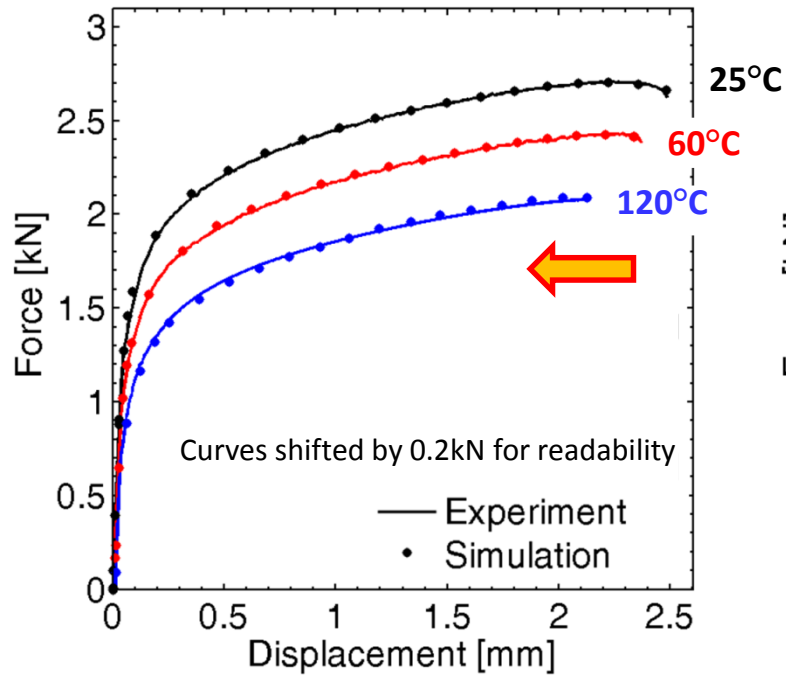
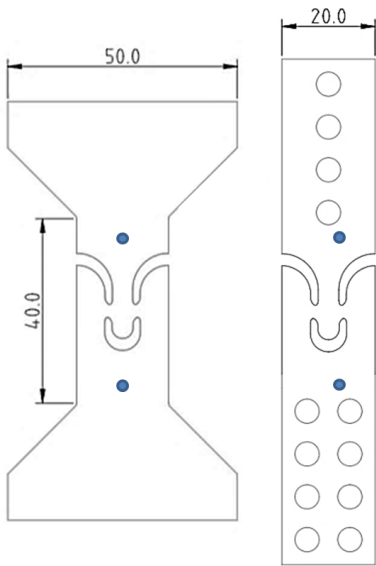
strain



ETH Tensile Testing at Elevated Temperatures

Slow @ 6.6 $\mu\text{m/s}$
(universal testing machine with
induction heating)

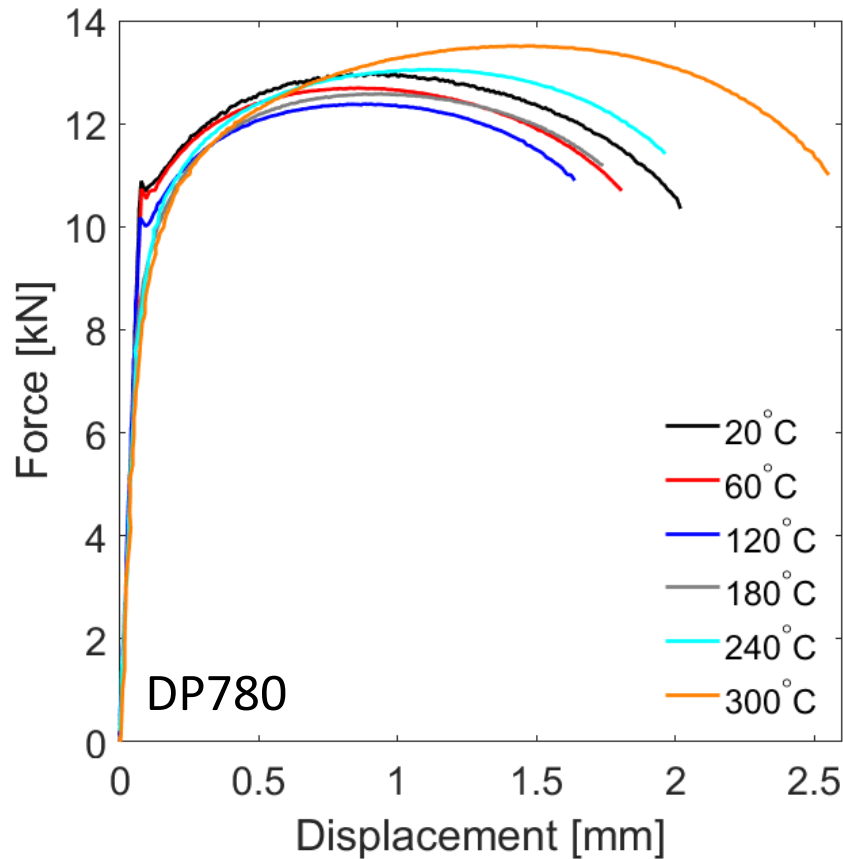
Fast @ 3.4m/s
(Hopkinson bar with
induction heating)



- **Non-monotonic effect of temperature on displacement to fracture**

Quasi-static Experiments from 20 to 300°C

Li, Roth and Mohr (2019)



Non-monotonic temperature response on plasticity!

➡ New plasticity model needed to calculate local fields in fracture specimens!

Machine-Learning Based Johnson-Cook Plasticity Model

Li, Roth and Mohr (2019)

- Johnson-Cook plasticity:

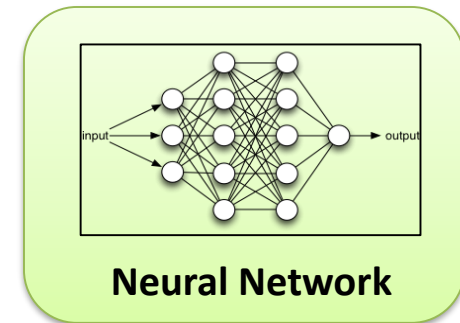
$$\sigma_y = k_1 [\overline{\varepsilon}_p] \times k_2 [\dot{\overline{\varepsilon}}_p] \times k_3 [T]$$

- New approach:

$$\sigma_y = k_1 [\overline{\varepsilon}_p] \times k_{NN} [\overline{\varepsilon}_p, \dot{\overline{\varepsilon}}_p, T]$$

Mixed Swift-Voce
strain hardening

Scaling factor for
temperature and
strain rate



Central Idea:

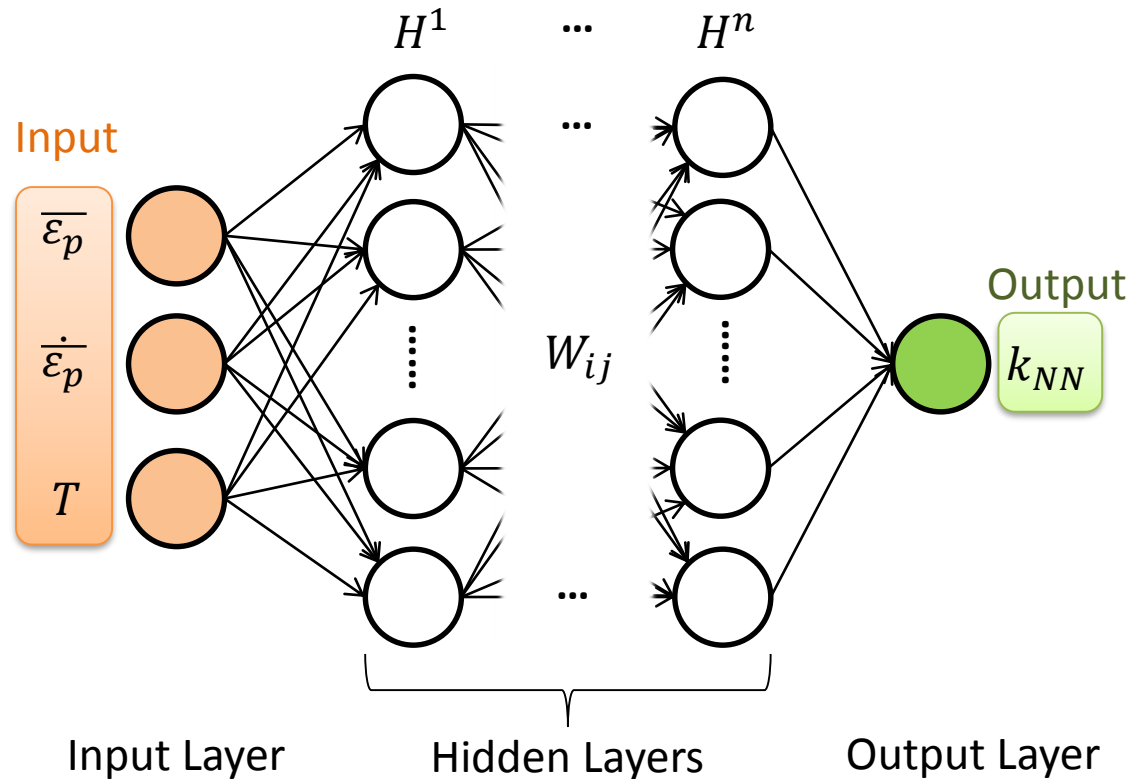
Introduce a neural network function to describe the effects of strain rate and temperature

Structure of Neural Network

Li, Roth and Mohr (2019)

$$\sigma_y = k_1 [\bar{\varepsilon}_p] \times k_{NN} [\bar{\varepsilon}_p, \dot{\bar{\varepsilon}}_p, T]$$

- $k_{NN} [\bar{\varepsilon}_p, \dot{\bar{\varepsilon}}_p, T]$ feedforward network with 10:10:10 structure

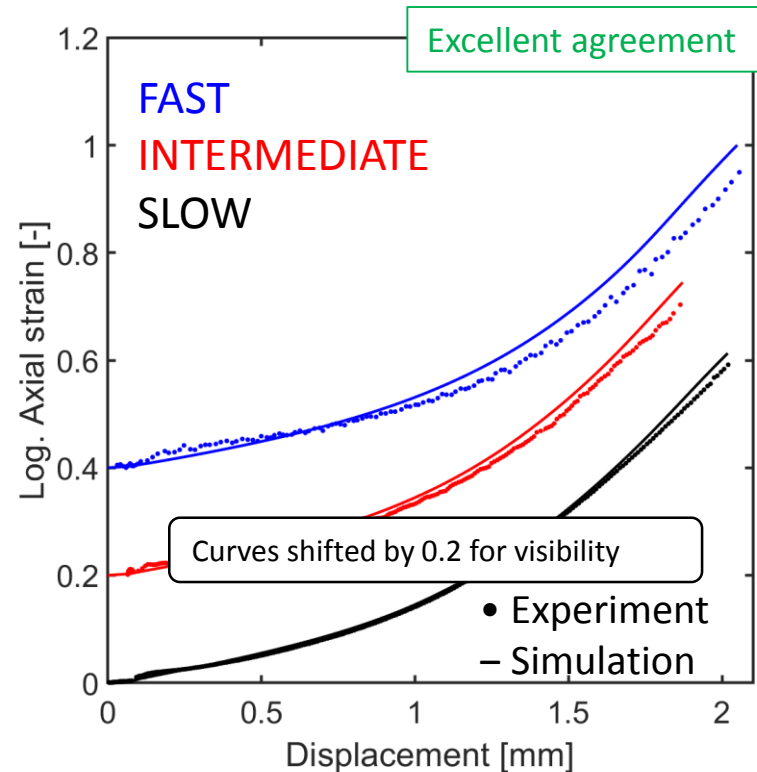
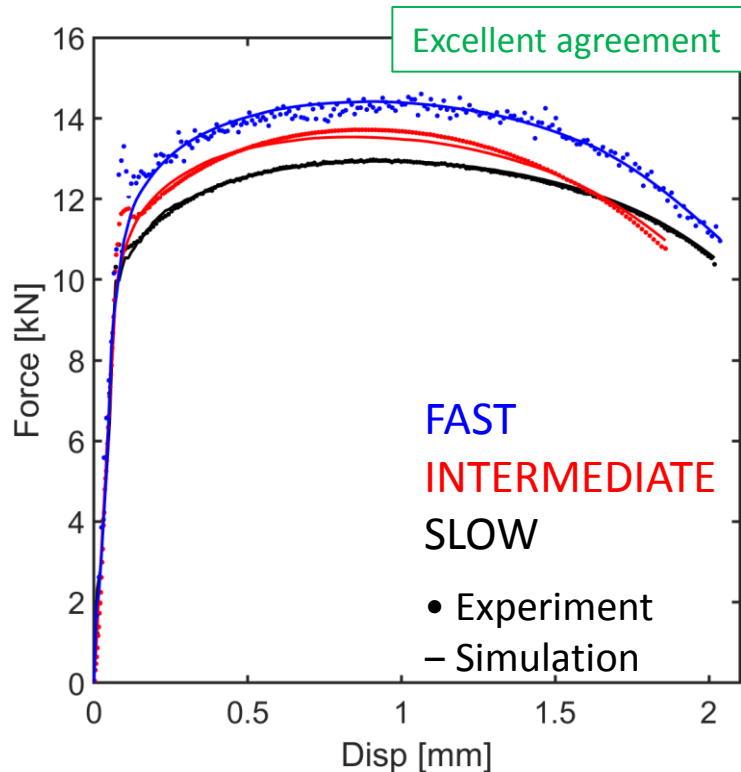


Implemented into user subroutine for Abaqus/Explicit

Performance of Trained Model

Li, Roth and Mohr (2019)

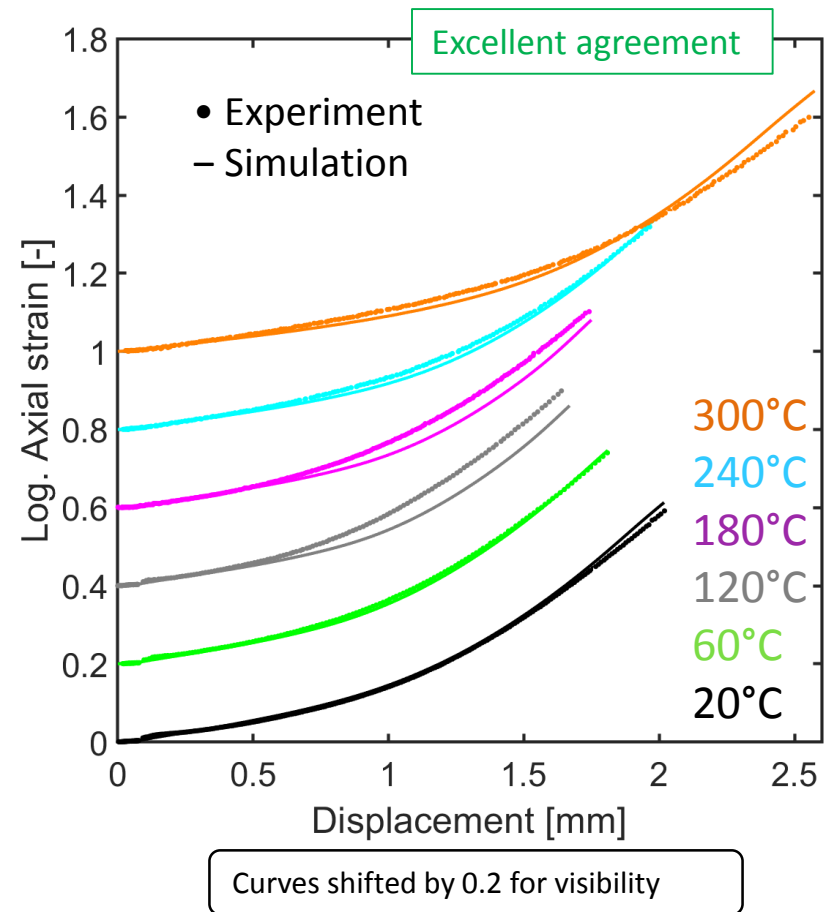
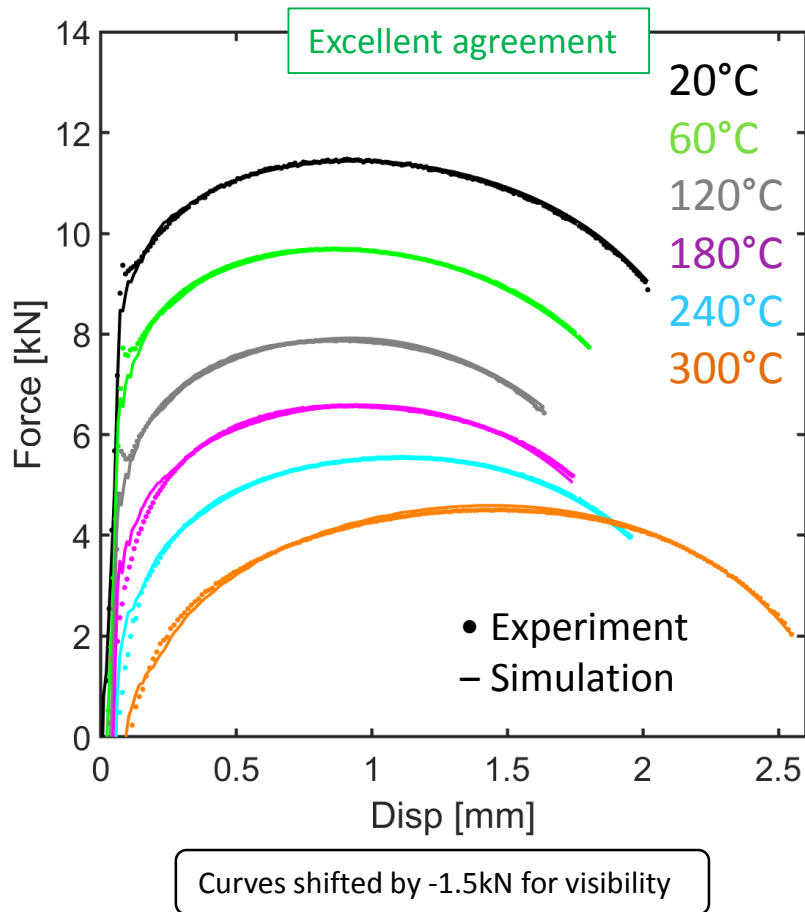
- Training data for loading @ room temperature (three experiments)



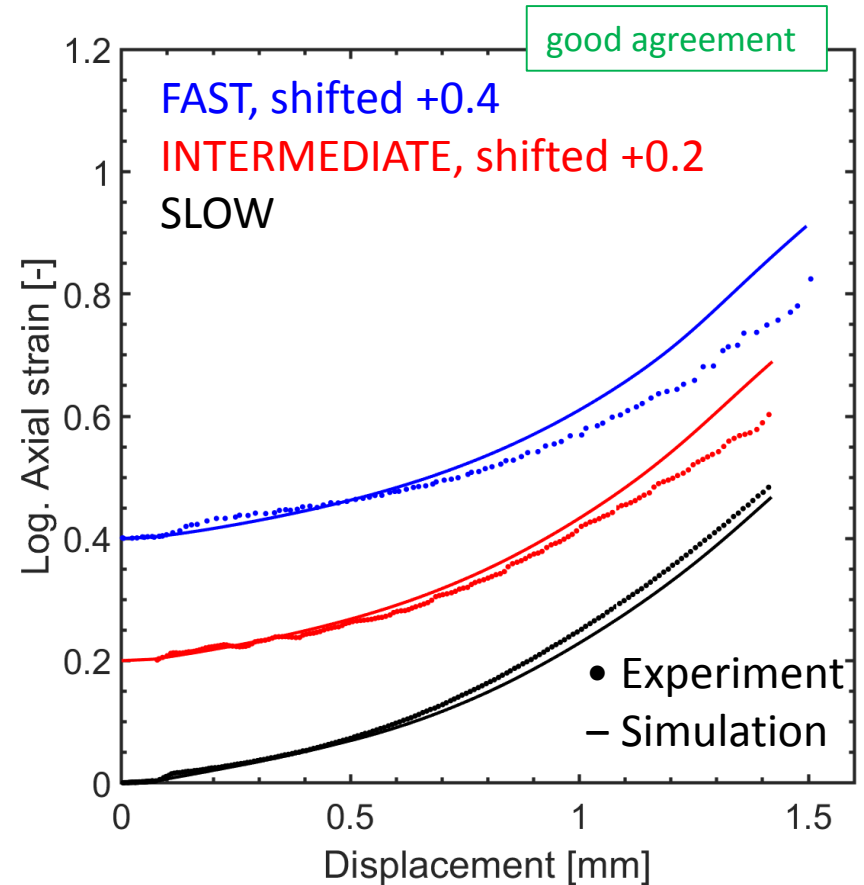
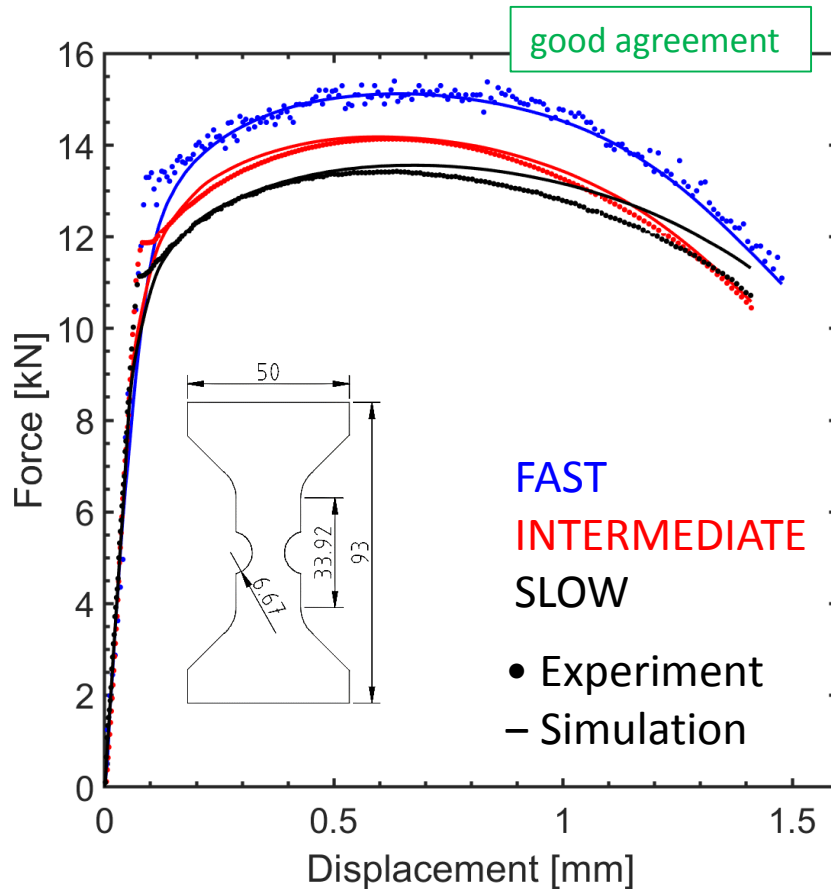
Performance of Trained Model

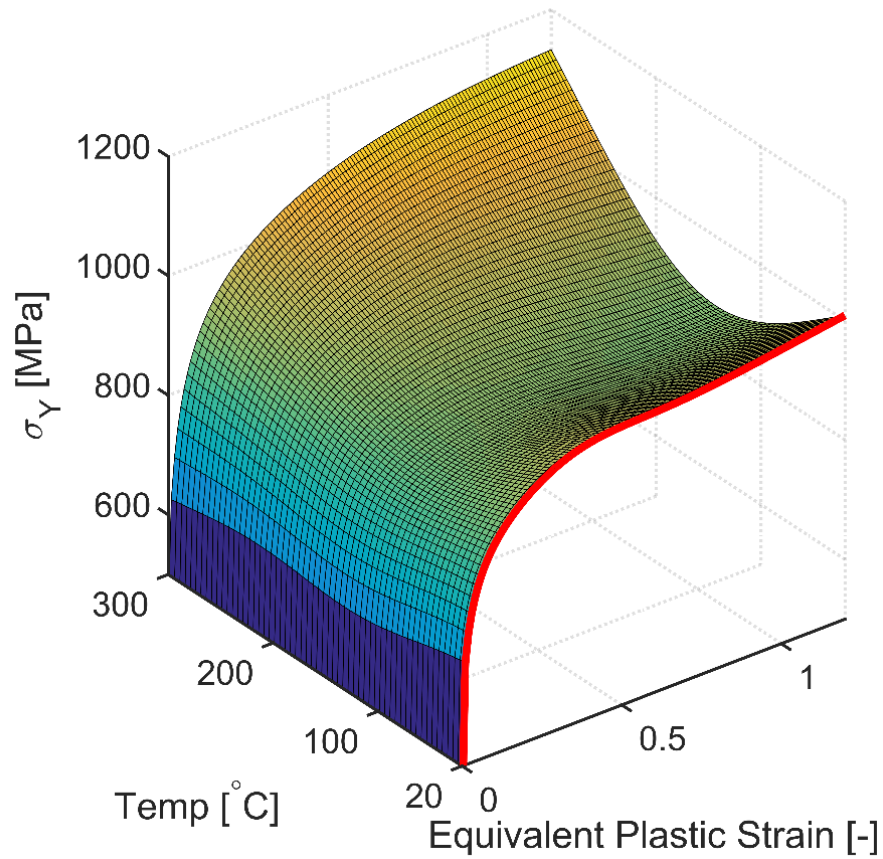
Li, Roth and Mohr (2019)

- Training data for quasi-static loading @ $\sim 10^{-3}/s$ (six experiments)

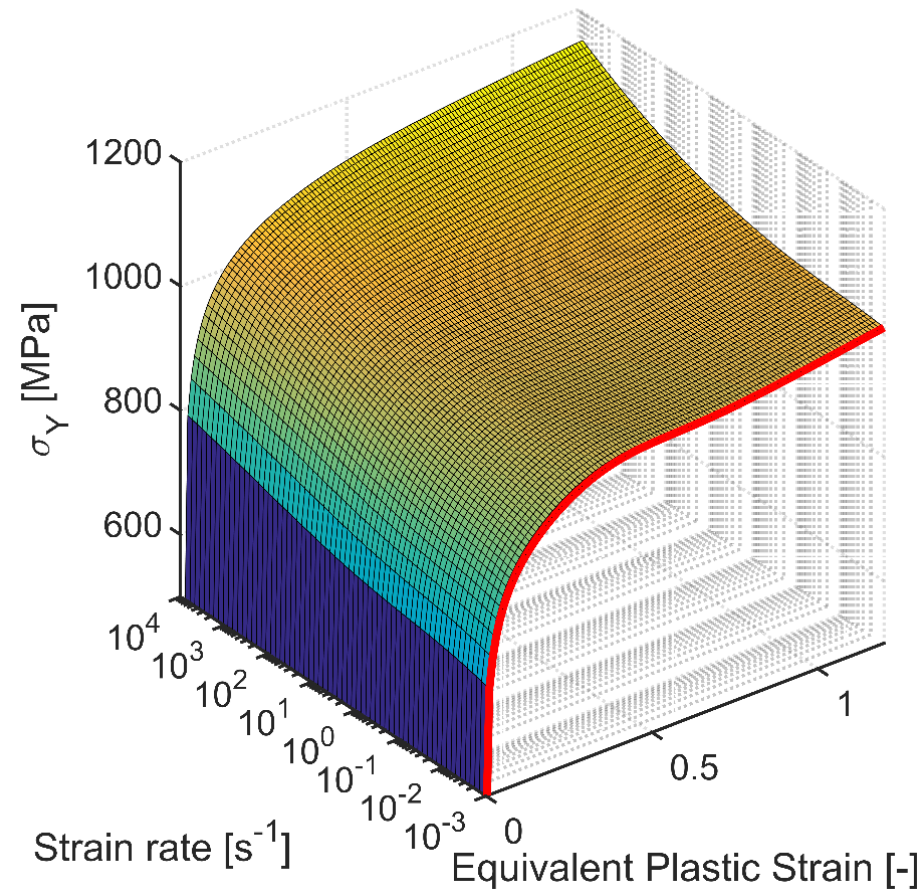


- Training data for loading @ room temperature (three NT6 experiments)



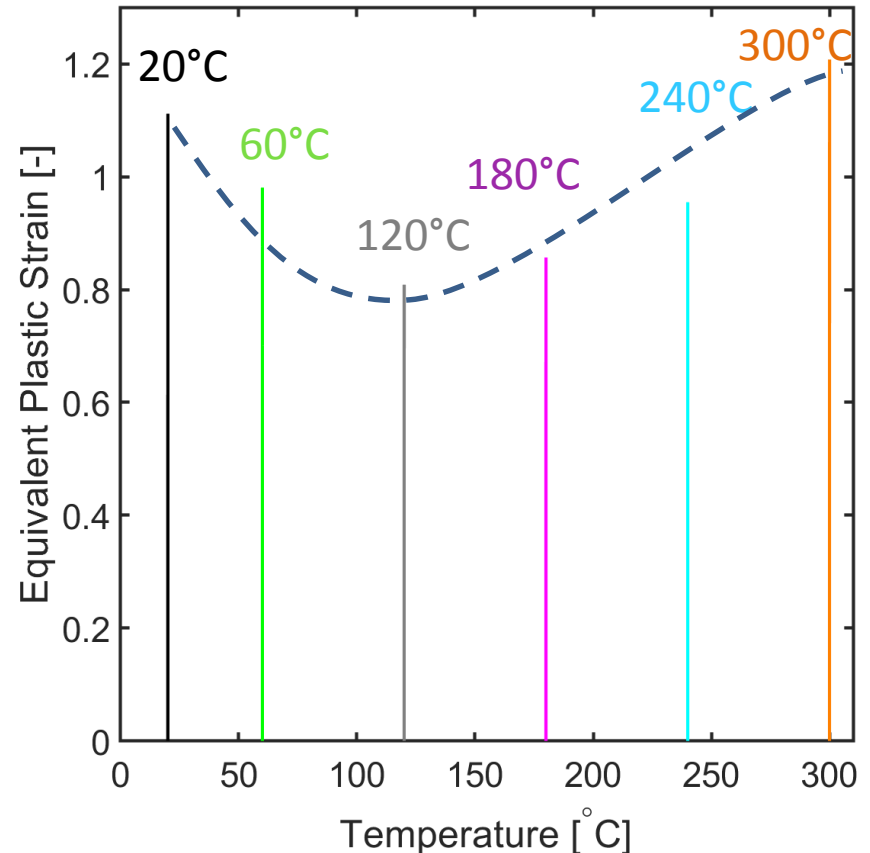
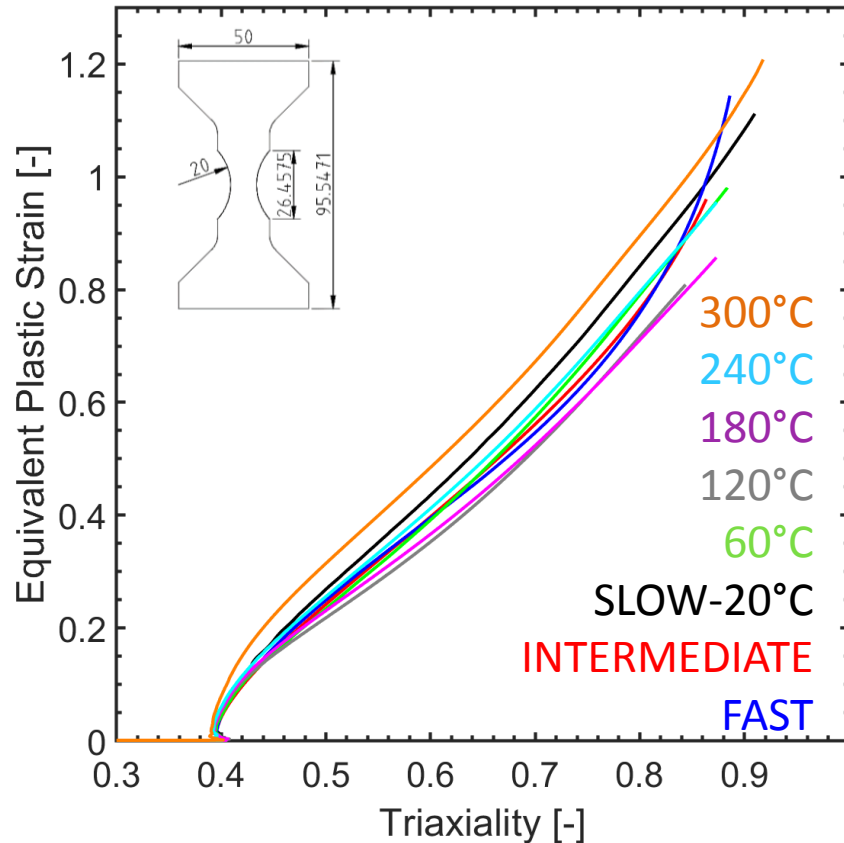
Quasi-Static ($10^{-3}/s$)

T=20°C



Loading Paths to Fracture

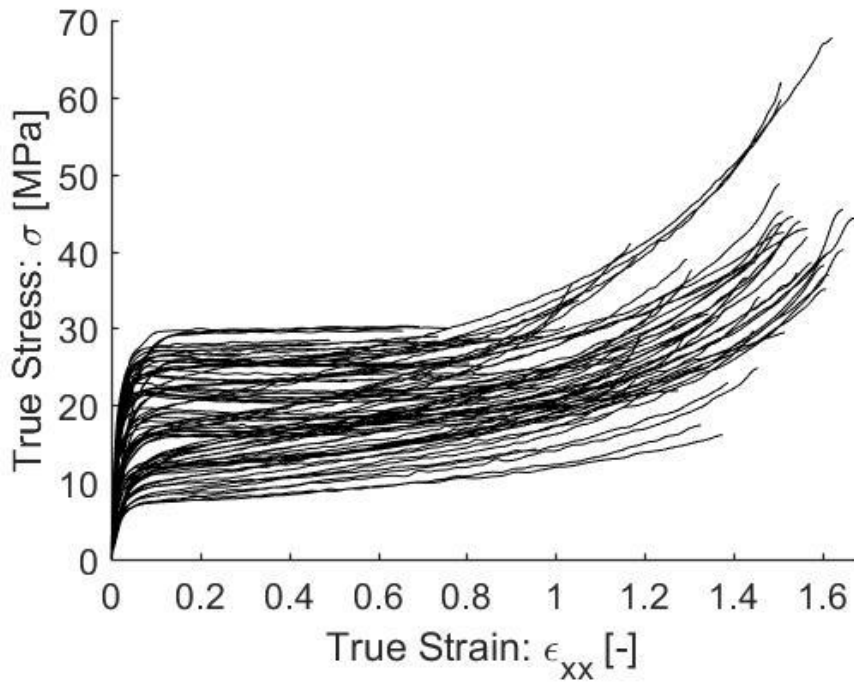
for quasi-static experiments on DP780 steel
Li, Roth and Mohr (2019)



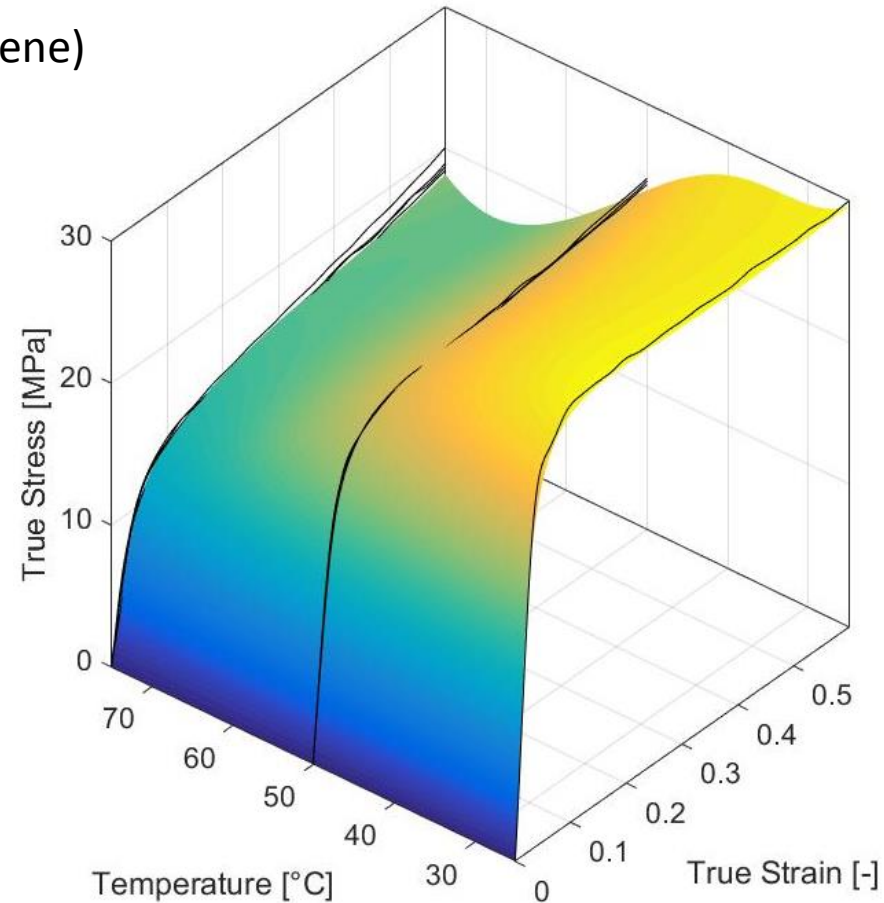
- **Non-monotonic effect of temperature on fracture strain!**

Jordan, Gorji and Mohr (2019)

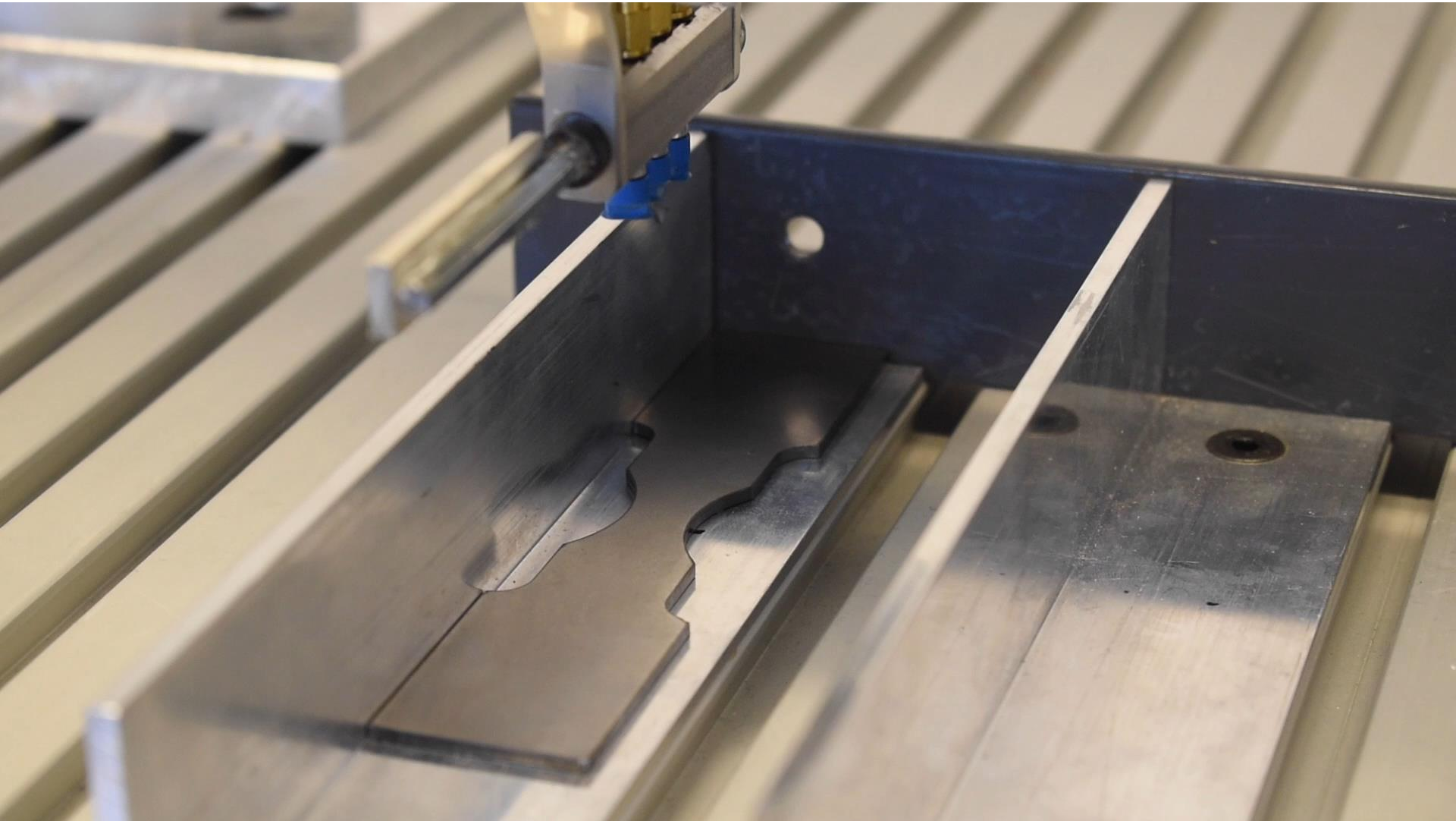
- Application to other materials (e.g. polypropylene)



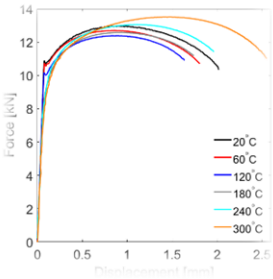
more than 100 experiments for
different temperatures and
strain rates



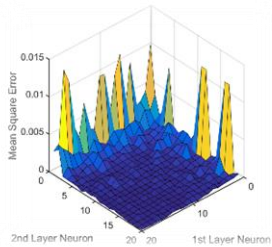
Machine-learning identified model



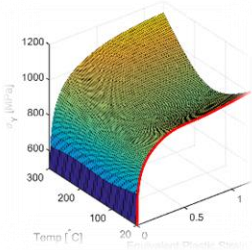
- Demonstrated non-monotonic effect of the temperature on the plasticity of dual phase steel



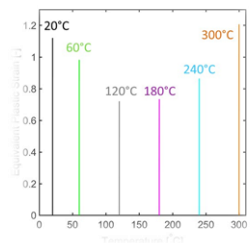
- Proposed a Neural-Network based temperature/strain rate term as a substitute of the classical Johnson-Cook term



- Implemented the model into material user subroutine of Abaqus/explicit, trained & validated the model



- Observed non-monotonic effect of the temperature on the fracture strain at RT



- **Important contributors**



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• Visiting Prof.



C. Roth
• Scientist



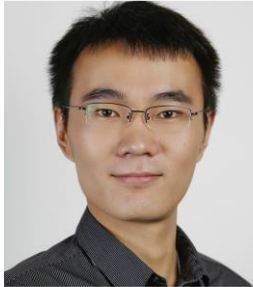
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B. Jordan
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